

1 **Areca nut husk biochar as a sustainable carbonaceous filler for cement:** 2 **Pyrolysis temperature and its effect on characterization, strength, and** 3 **hydration**

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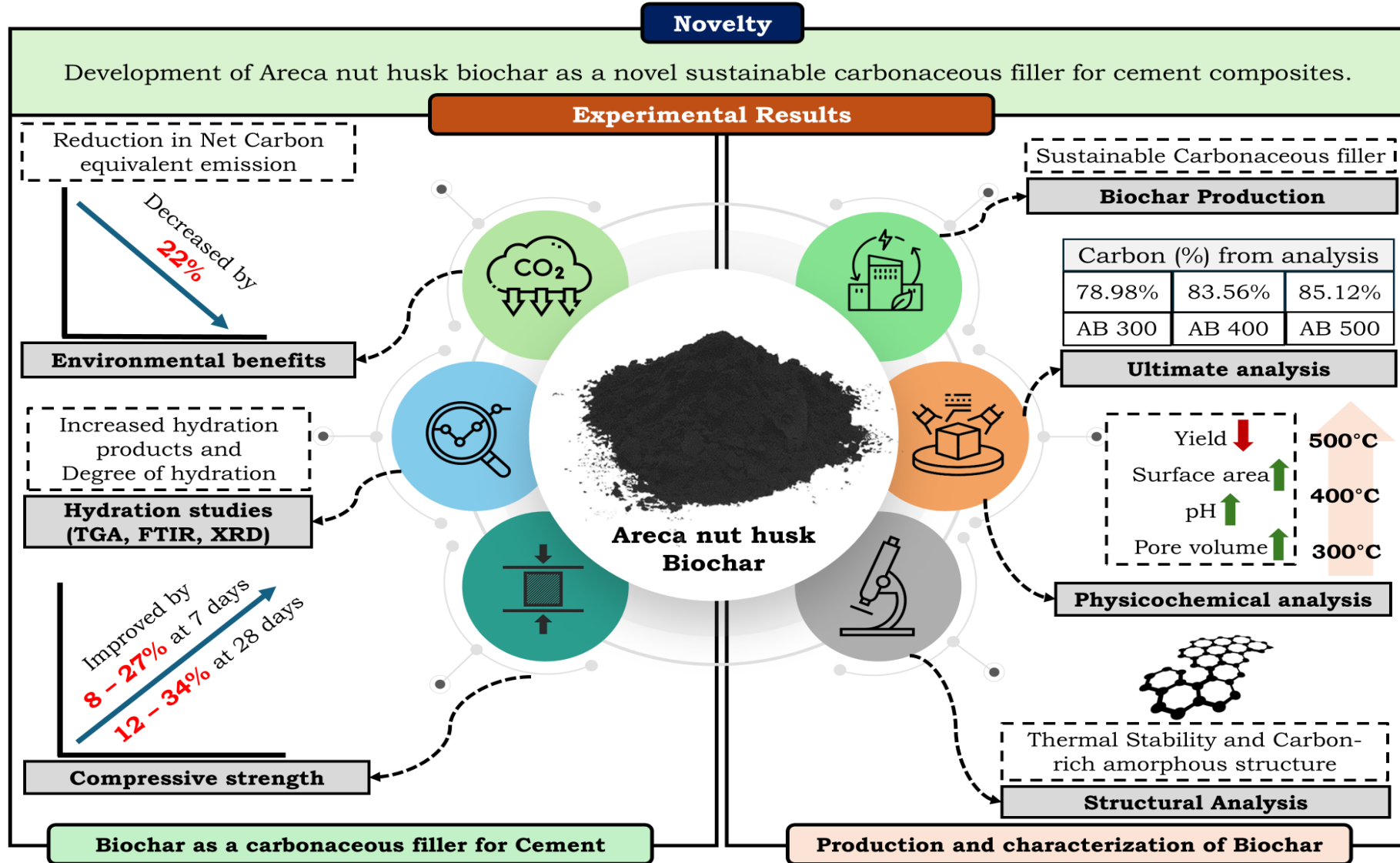
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16 **Highlights**

- 17 • Areca nut husk is a potential agro-waste for the preparation of biochar, offering a sustainable
18 carbonaceous filler for cement.
- 19 • Increase in the pyrolysis temperature of areca nut husk biochar (AB) leads to a decrease in
20 H/C and O/C, indicating the aromaticity of the biochar.
- 21 • Micro filler effect of AB improves the early strength of cement mortar.
- 22 • Degree of hydration is improved with the addition of AB, confirmed by Bhattya's method.
- 23 • AB reduced CO₂ equivalent emissions compared with cement mortar.

Graphical Abstract



Abstract

27 This study addresses the gap in sustainable agro-based materials for cement by exploring
28 locally available areca nut husk pyrolyzed into areca nut husk biochar (AB). The research
29 investigated the effect of pyrolysis temperature (300°C, 400°C, and 500°C) on the
30 characteristics of AB and its impact on cementitious performance. The study found that
31 increasing pyrolysis temperatures led to lower yield, greater aromaticity, and increased surface
32 area of AB. Fourier Transform Infrared Spectroscopy (FTIR) analysis showed decreased
33 functional groups in AB at higher temperatures, confirming enhanced carbonization.
34 Thermogravimetric analysis (TGA) revealed greater thermal stability of AB. X-ray diffraction
35 (XRD) indicated a carbon-rich amorphous structure and crystalline graphite carbon formation
36 in AB. Incorporating AB at 2% into cementitious composites substantially increased the
37 compressive strength compared to the control mortar. At 7 and 28 days, the compressive
38 strength increased by 8% and 12% for AB 300, 16% and 21% for AB 400, and 27% and 34%
39 for AB 500. This improvement was due to the micro filler effect of AB, which improved the
40 compactness of the cementitious matrix. Hydration studies from TGA showed that the addition
41 of AB accelerated early-stage hydration, with the degree of hydration increasing from 46% (in
42 control mix) to 48-53% in AB blended mixes using Bhatti's method. FTIR analysis
43 demonstrated improved hydration of silicate phases and C-S-H formation in the presence of
44 AB, supported by XRD analysis. AB blended mortar reduced the CO₂ equivalent emission by
45 22% compared to the control mortar attributed to its carbon sequestration capacity. These
46 results highlight the potential of AB as a sustainable carbonaceous filler for cementitious
47 composites, offering an environmentally friendly option for future research in construction
48 materials.

49 **Keywords:** Areca nut husk biochar, Carbonaceous filler, Strength, Hydration, Sustainability

50 **1. Introduction**

51 The significant increase in global carbon dioxide (CO₂) emissions and other greenhouse
52 gases (GHG) severely impacted the environment (Liu et al., 2019), leading to climate change
53 and ecological imbalances. Anthropogenic activities threaten the ecosystems and stability of
54 society, endangering the nation (Selvakumar et al., 2022). According to the Intergovernmental
55 Panel on Climate Change (IPCC), by the end of the 21st century, atmospheric CO₂
56 concentrations will rise from 410 to 590 ppm, causing a potential global temperature to increase
57 by 1.9°C (Marescaux et al., 2018). Numerous countries have set a goal to achieve carbon
58 neutrality to counteract global warming. The European Union, China, and India have set targets
59 for achieving carbon neutrality by 2050, 2060, and 2070. Several nations are implementing
60 carbon emissions trading schemes based on market principles to incentivize a reduction in
61 carbon dioxide emissions for those who surpass approved thresholds (Chen and Lin, 2021).

62 The construction sector is a significant contributor to GHG, accounting for
63 approximately 40% of all energy CO₂ emissions (Rashid et al., 2024). Within this sector, the
64 production of Portland cement is responsible for about 36% of the 7.7 gigatons of
65 anthropogenic CO₂ emissions released globally by construction activities (Habert et al., 2020).
66 This makes cement production the second-largest industrial source of CO₂ emissions
67 worldwide, each ton releasing approximately 800 kg of CO₂ into the atmosphere (Andrew,
68 2018).

69 On the other hand, agricultural waste management presents a significant challenge due
70 to its low recycling rate and substantial environmental impact. Agricultural activities generate
71 approximately 1 billion tons of waste annually, contributing to about one-fifth of total GHG
72 worldwide (Karić et al., 2022). In India, the Indian Council of Agricultural Research reports an
73 annual generation of around 350 million tons per year (Thakur et al., 2020), which is expected
74 to rise due to population growth (Lee et al., 2022). The current disposal methods, such as
75 landfilling or incineration, account for 3% of global GHG emissions (Khan et al., 2021). The
76 uncontrolled burning of agricultural waste further poses environmental concerns. Given these
77 challenges, there is an urgent need to develop sustainable management practices for agricultural
78 waste. Consequently, researchers are increasingly focusing on innovative approaches to reuse
79 this waste, particularly in the development of green and low-carbon construction materials.

80 Carbon sequestration in cement-based materials offers an effective strategy to reduce
81 CO₂ emissions in the construction industry (Javed et al., 2022). All agricultural wastes with
82 high residual carbon content and low cost can be used to convert into porous carbon materials,

83 called biochar (Kim et al., 2023; Zhang et al., 2020). In this context, biochar emerges as an
84 innovative solution to address environmental pollution and global warming, owing to its unique
85 physicochemical properties and diverse applications. Biochar is a carbon-rich aromatic
86 substance derived from biomass through thermochemical conversion under oxygen-limited
87 conditions (Lehmann et al., 2011). Biochar acts as a carbon sequestration agent in various
88 applications such as soil amendment, soil health, nutrient retention, soil pH, energy production,
89 and building material. The IPCC reports biochar as a promising strategy to promote carbon
90 neutrality, as one ton of biochar can sequester up to 2.6 tons of CO₂ (Azzi et al., 2019).
91 Furthermore, biochar produced from biomass shows significant potential in the development
92 of sustainable carbon materials (Yu et al., 2022). The production and characteristics of biochar
93 are influenced by various factors illustrated in Fig. 1.

94 Several studies have explored the potential of storing and fixing carbon in soil with
95 biochar. The properties of biochar are beneficial for both soil amendment and construction
96 purposes. The addition of biochar into cement reduces the landfill demand, prevents the
97 burning of waste, and promotes waste utilization (Sirico et al., 2021). Incorporating 1 kg of
98 biochar into concrete can sequester approximately 2.5kg of CO₂ emissions (Dixit et al., 2021).
99 The addition of biochar to cementitious mixtures is a promising attribute as a green admixture,
100 making it an ideal option for concrete applications. Recent studies have demonstrated that
101 biochar in cementitious materials can significantly enhance mechanical properties. Biochar
102 produced at lower temperatures (300°C - 500°C) generally showed optimal results, as
103 evidenced by mixed wood sawdust pyrolyzed at 300°C (2% replacement), improving early
104 compressive strength (Gupta et al., 2018c), while in another study, biochar prepared with
105 sawdust at 500°C showed a 45% improvement in strength (Ali et al., 2023). Further, waste
106 wood biochar pyrolyzed at 400°C and 500°C (1 – 3% replacement) increased the strength by
107 25% (Tan et al., 2020a). Wood-based biochar proved to be very effective with pre-soaked wood
108 sawdust (2% replacement) exhibiting an increase in compressive strength results by 40 – 50%
109 (Gupta and Kua, 2018). The performance of biochar is significantly influenced by its source
110 material, and agricultural waste-derived biochar has demonstrated promising results. Rice husk
111 and bagasse biochar produced at 700°C (5% replacement) increased compressive strength by
112 36% and 55%, respectively (Zeidabadi et al., 2018). Olive-derived biochar has shown potential,
113 with olive stone biochar (4% replacement) slightly improving compressive strength (Maljaee
114 et al., 2021b) and olive tree pruning biochar at 5% replacement, resulting in a 13%
115 improvement in strength (Kalderis et al., 2024). Other agricultural wastes like coconut, peanut,
116 and wheat husk biochar (Javed et al., 2022), and jungle keekar (Rashid et al., 2024) at 2% and

117 5% of the cement replacement enhanced the strength. Industrial and municipal waste biochar
118 have also been explored, with pulp and paper mill sludge biochar at 0.1% replacement
119 providing similar strength to the control specimen (Akhtar and Sarmah, 2018). Nano-biochar
120 from municipal solid waste at 0.12% replacement improved strength by 17% (Sisman et al.,
121 2024). Various studies have shown that accelerated carbonation curing for biochar improved
122 the mechanical properties of cementitious composites irrespective of the feedstock used (Kua
123 and Tan, 2023; Yang and Wang, 2021). The optimal biochar dosage varies across studies but
124 generally falls within the range of 1-5% by weight of cement. These comprehensive findings
125 of biochar from diverse sources, along with different application methods, pyrolysis
126 temperature, residence time, and quantity, exhibit biochar's significant potential in developing
127 sustainable construction materials.

128 While existing research has explored a wide range of feedstocks for biochar
129 preparation, there remains unexplored potential in other agricultural wastes. Notably, the use
130 of areca nut husk biochar (AB) in construction applications represents a novel approach that
131 has not been previously explored in the literature. This study aims to discover new avenues for
132 sustainable construction practices by investigating the effects of AB as a sustainable
133 carbonaceous filler in cementitious composites. The utilization of AB adds to the diversity of
134 biochar sources and offers region-specific solutions, particularly in areas where areca nut waste
135 is abundant. This novel application of AB in construction materials emphasizes the ongoing
136 investigation in the field and the continuous efforts to find sustainable alternatives in the
137 construction industry.

138 Areca nut, or betel nut, is a fruit of areca catechu palm species grown extensively in
139 Southeast Asian countries (Bera and Mohanty, 2020). India is the world's largest producer of
140 areca nut, with approximately 904 tons, with the majority of production occurring in Karnataka
141 (Hugar et al., 2023). The areca nut husk is a lignocellulosic biomass that constitutes about 65
142 – 80 % of the total weight of the fruit (Vikraman et al., 2022). The areca nut is covered by a
143 fibrous husk, discarded as agricultural waste with an estimated around 6 to 7 tons of waste
144 available from 1 ha of areca nut plantation yearly, and has no market value. Improper disposal
145 techniques of areca husks have led to significant environmental problems (Anuar et al., 2021).
146 It undergoes slow decomposition and, therefore, cannot be used as manure. Fig. 2 shows the
147 areca nut dumped (DHNS, 2020).

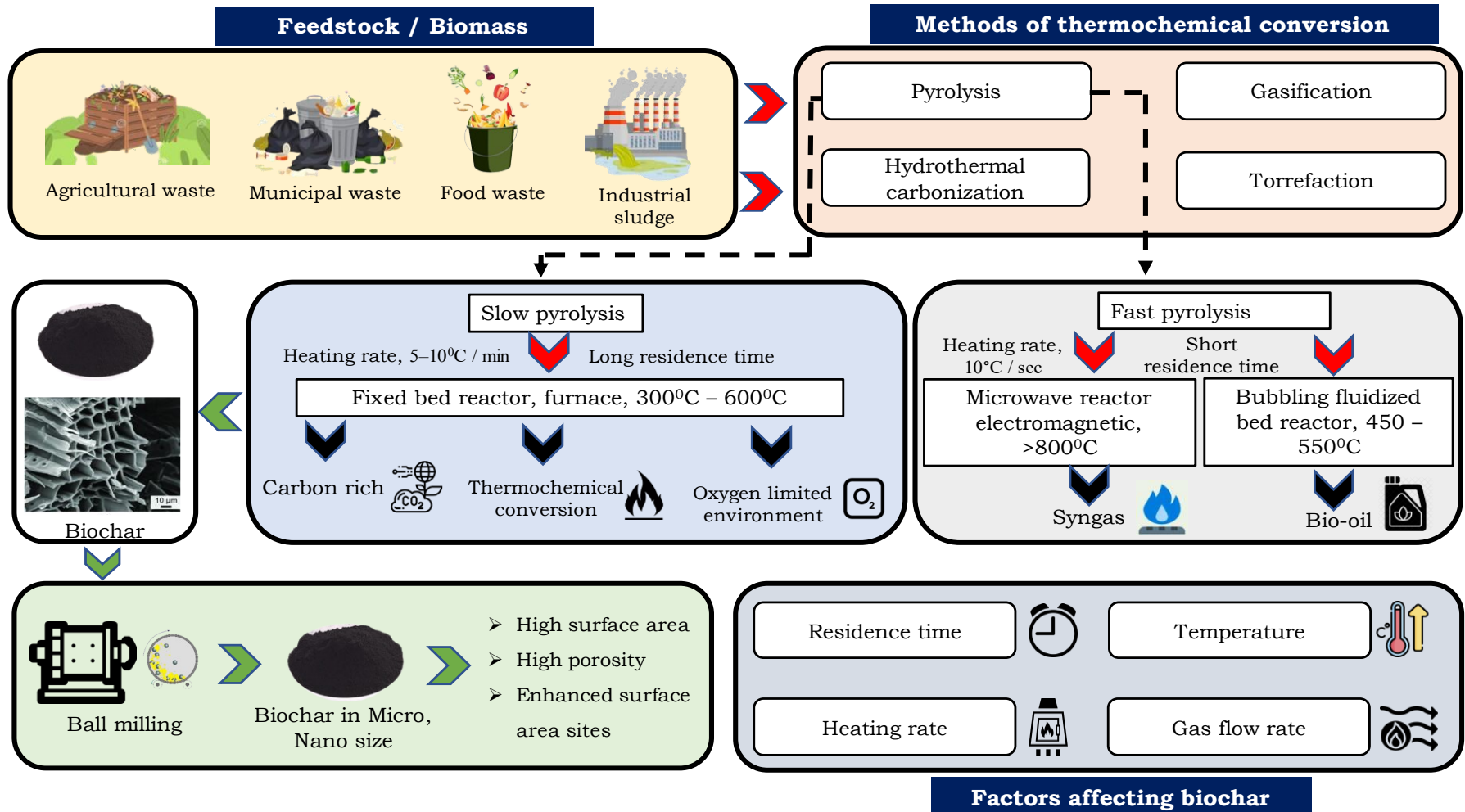


Fig. 1. Different feedstocks and methods to produce biochar



Fig. 2. Arecanut husk dumped

151 Few studies have shown that AB has the potential to remove heavy metals and soil amendment,
152 improve fuel properties, and act as an adsorption agent. However, its use as a building material
153 is still in its early stages. Hence, this study introduces an AB to the scientific community,
154 expanding upon the conventional biochar utilized in cementitious composites. Further
155 investigation and verification of the utilization of AB in engineering applications are needed to
156 promote its wider adoption as a building material. The objectives of the present study are as
157 follows:

- 158 1. To investigate the in-depth characteristics of novel green AB at different pyrolysis
159 temperatures.
- 160 2. To determine the feasibility of AB as a sustainable carbonaceous filler in
161 cementitious composites.

162 The present study carried out a comprehensive experimental program to achieve these
163 objectives. Proximate analysis, ultimate analysis, physicochemical analysis, surface analysis,
164 and structural analysis are carried out to investigate the characteristics of novel AB.
165 Compressive strength, ultrasonic pulse velocity (UPV), and hydration studies such as X-ray
166 diffraction (XRD), thermogravimetric analysis (TGA), and Fourier transform infrared
167 spectroscopy (FTIR) were conducted.

168 **2. Materials and Methods**

169 **2.1 Raw ingredients used**

170 Ordinary Portland Cement (OPC) of grade 53 manufactured from UltraTech Cement Pvt. Ltd
171 was used for all experimental work. The chemical composition and physical properties of the
172 cement are presented in Table 1 and Table S1. The particle size distribution of the OPC was
173 determined using a laser diffraction particle size analyzer, as shown in Fig. 3. The D10, D50,
174 and D90 values of OPC are 3.12 μm , 22.49 μm , and 57.80 μm , with the mean particle size of
175 27.40 μm . The specific surface area of OPC is 0.34 m^2/g . This study used locally available
176 natural river sand as fine aggregates. The sand was washed to remove fine substances, sun-
177 dried, sieved, and graded. The particle size curve for river sand (Fig. 3) confirms zone – II. The
178 fine aggregate has a specific gravity of 2.68. The physical properties of sand are summarized
179 in Table S2. Potable tap water was used to maintain the required moisture content during the
180 casting and curing stages.

181 **Table 1 Chemical composition of OPC**

Chemical Composition (%)	SiO ₂	CaO	Al ₂ O ₃	Fe ₂ O ₃	MgO	SO ₃	K ₂ O	Na ₂ O	LOI
	18.59	63.87	6.22	4.98	1.54	3.24	0.25	0.2	0.98

182 **2.2 Biochar Production**

183 Areca nut husk, from the matured outer cover of an areca nut, was procured from Mangaluru,
184 Karnataka, and was subjected to sun drying before further processing. The husks were then
185 dried in the oven at 100°C for 24 h. The dried husks are further processed by mechanically
186 reducing the size of the husk in a ball mill for 1 h and kept in the desiccator before being used
187 to produce biochar.

188 Fig. S1 shows a programmable retort furnace used to produce biochar. The samples
189 were subjected to pyrolysis at three different temperatures - 300°C, 400°C, and 500°C with a
190 heating rate of 10°C / min and 1 h residence time. The furnace was purged with nitrogen gas
191 to eliminate volatile products and ensure an oxygen-free atmosphere. The biochar produced by
192 pyrolysis is shown in Fig. 4. The produced biochar was allowed to cool down to room
193 temperature in a nitrogen environment. Fig. 5 shows the areca nut husk and the produced AB.
194 The solid biochar remaining in the furnace was weighed and ground to a finer size for 3 min in
195 a mixer grinder to AB and stored in the desiccator for characterization.

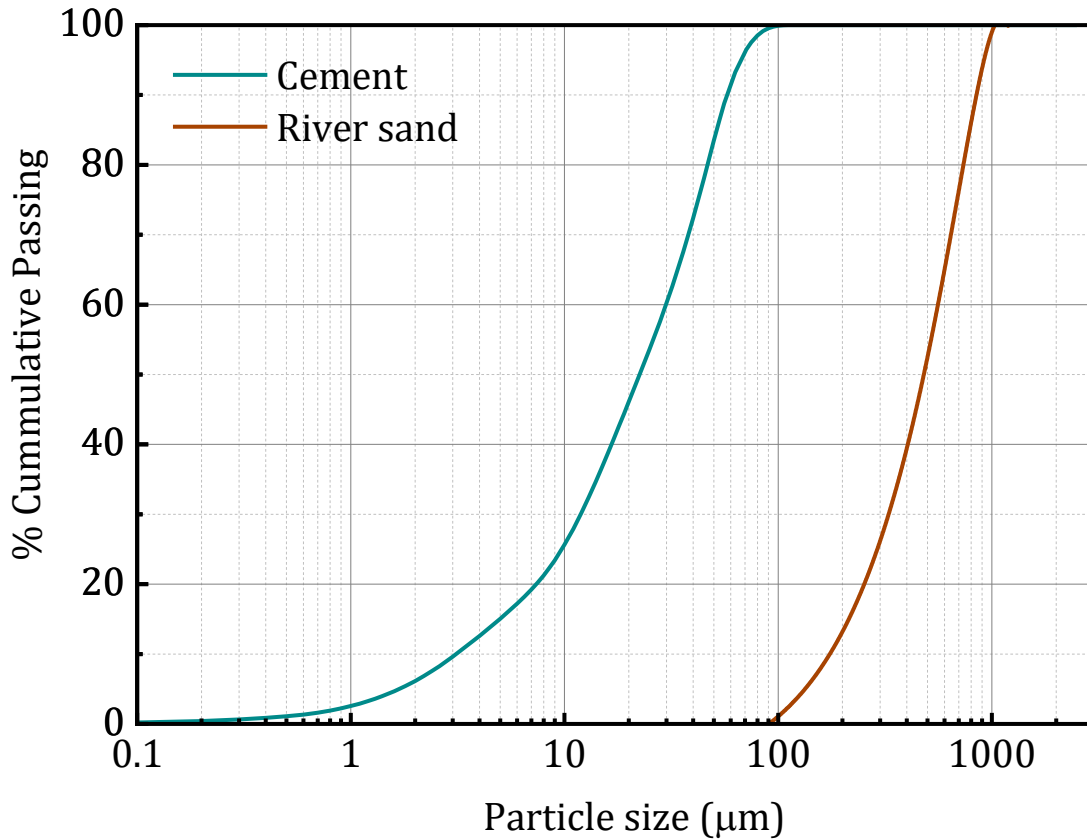


Fig. 3. Particle size distribution of raw materials

196 **2.3 Characterization of biochar**

197 The characterization techniques used for the produced AB are shown in Fig. 6.

198 **2.3.1 Proximate analysis**

199 Proximate analysis was conducted to assess critical parameters in AB, including moisture
 200 content (MC), volatile matter (VM), ash content (AC), and fixed carbon (FC), following ASTM
 201 E871-72, ASTM E872-82, and ASTM D1102-84, respectively. The comprehensive results of
 202 the proximate analysis are presented and discussed in section 3.

203 **2.3.2 Ultimate analysis**

204 The ultimate analysis of AB was performed to determine the percentages of Carbon (C),
 205 Hydrogen (H), Sulphur (S), Nitrogen (N), and Oxygen (O). The elemental model vario EL III
 206 elemental analyzer employs helium as a carrier gas combusted with approximately 10 mg of
 207 AB at 1100°C. The resulting gases (CO₂, H₂O, NO₂) were then transported to a detector to
 208 quantify the percentages of each element.

209 **2.3.3 Physico-chemical analysis**

210 The specific surface area, pore size, and pore volume of AB were determined using BET
 211 (Brunauer-Emmet-Teller) N₂ adsorption on the Autosorb IQ instrument by Anton Paar.
 212 Approximately 50 mg of AB was degassed for 8 h at 105°C. Particle size and zeta potential

213 were measured using dynamic light scattering (DLS) (Gujre et al., 2022) with the Lifesize 500
214 instrument. To determine pH values, AB samples were added to de-ionized water in a 1:20
215 mass ratio, followed by manually agitating and allowed to settle for 5 min before measuring
216 with a pH meter (Gogoi et al., 2017). The same suspension is used to determine electrical
217 conductivity using a conductivity meter.

218 **2.3.4 Surface analysis**

219 The surface morphology of AB samples was examined using a scanning electron microscope
220 (SEM) (Joel, Japan).

221 **2.3.5 Structural analysis**

222 Crystalline phases present in AB were studied using XRD. The samples were scanned in the
223 2θ range of 5 - 80° at a scan speed of 0.02°/s, from which qualitative information is extracted
224 and analyzed. TGA characterizes the phase composition of the cement paste (Bhojaraju et al.,
225 2021). TGA was performed on AB samples using a PerkinElmer thermal analyzer (TGA 4000).
226 The weight loss was then monitored from 25 to 900°C, with a heating rate of 10°C / min under
227 a protective nitrogen atmosphere. For FTIR analysis, the samples of AB were mixed with KBr,
228 and these samples were then made into pellets using a pelletizer. The resulting pellets were
229 placed in an FTIR instrument (PerkinElmer), which functioned from 600 to 4000 cm^{-1} to
230 analyze the chemical groups contained in the samples.

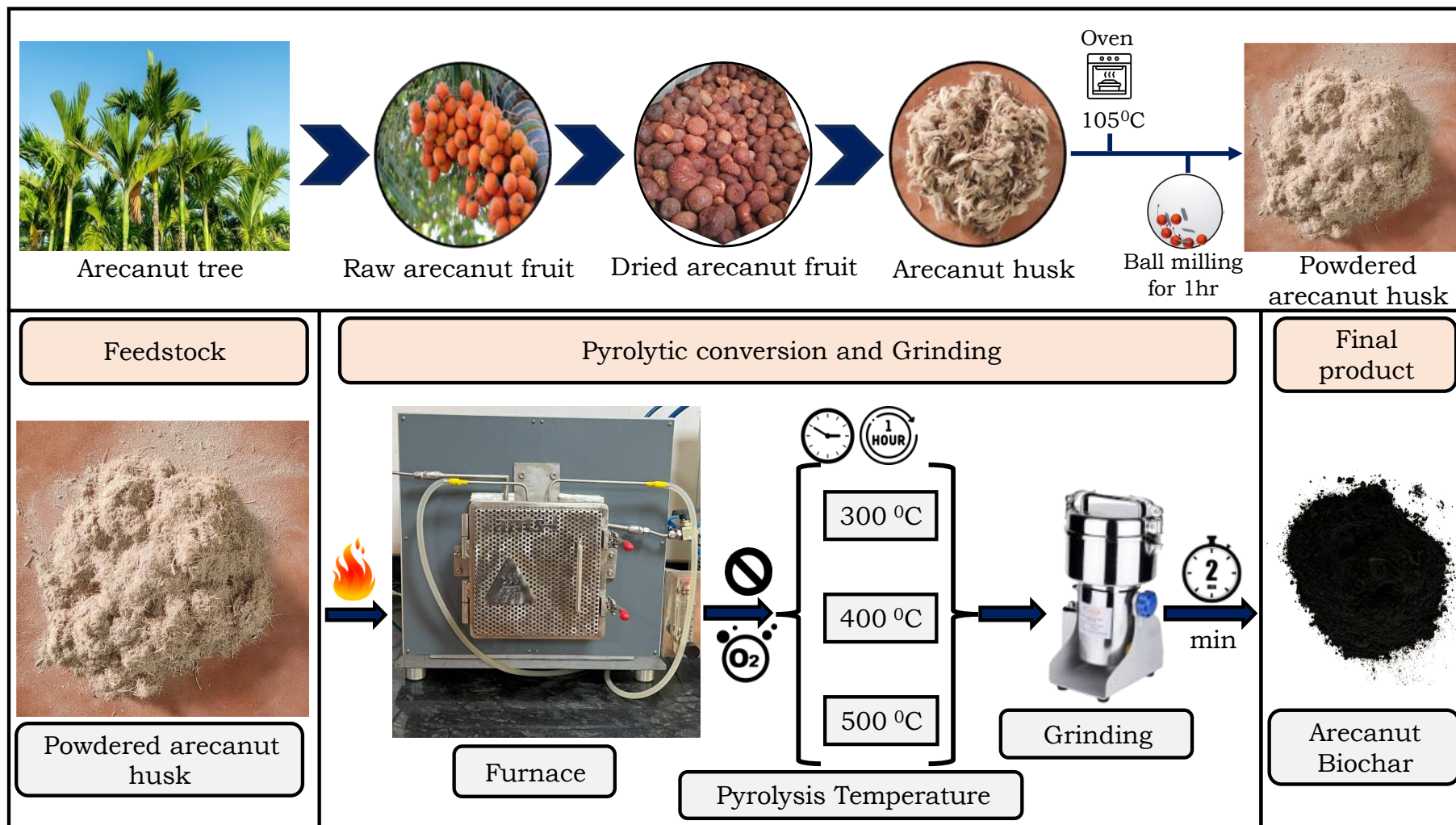


Fig. 4. Process used for the production of AB



Fig. 5. a) Dried areca nut husk – Before pyrolysis b) AB – After pyrolysis

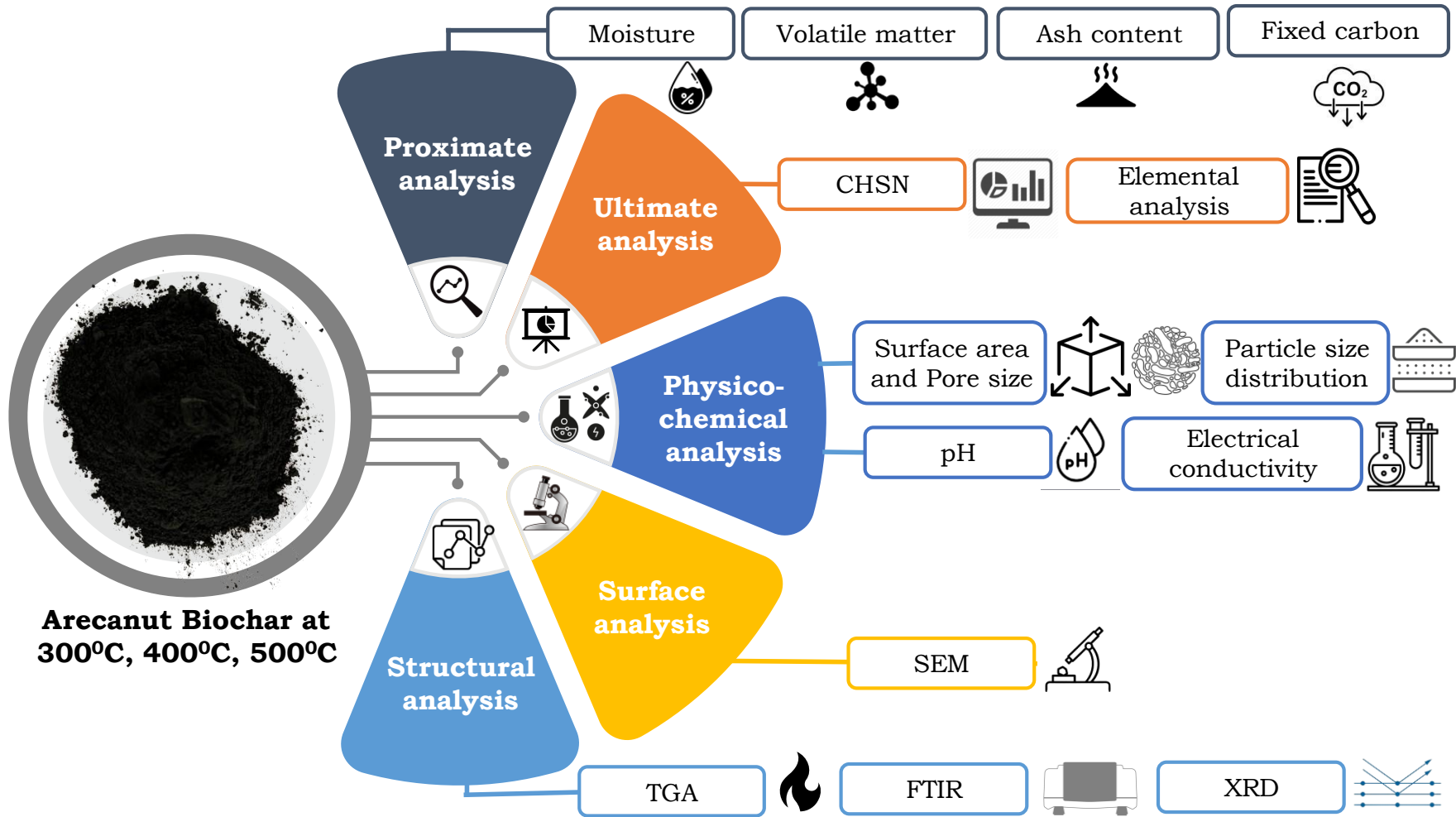


Fig. 6. Characterization techniques to determine the properties of biochar

234 **2.4 Mix proportion and specimen preparation**

235 The mortar was prepared with a proportion of 1:2.5:0.40 by weight of cement, sand, and water.
236 The AB was mixed into cementitious composites at 2% by weight. The details of the mix
237 proportions are shown in Table 2. The dry materials were mixed manually, and then the mixing
238 water was added to the dry mixture. To attain homogeneity, all the materials are mixed at low
239 speed for 1 min and 30 sec at high speed, and the mortar is left to rest for 90 sec to clean the
240 sides of the bowl. The material is mixed for 1 min at high speed (Manjunath et al., 2023). After
241 mixing, the fresh mixture was poured into the steel moulds with a cube size of 50 mm. The
242 cubes were covered with polythene sheets for 24 h before being demoulded and then immersed
243 in water for 7 and 28 days of curing.

244 **Table 2** Mix proportions of paste and mortar samples

Mixes	OPC (g)	AB (g)	Water (g)	Sand (g)
PC	400	-	160	1000
AB 300	392	8	160	1000
AB 400	392	8	160	1000
AB 500	392	8	160	1000

245 **2.5 Experimental methods**

246 The study tested compressive strength tests on both control cement and AB-blended cement
247 mortar at 7 and 28 days as per ASTM C-109. Three samples of each mix were tested for
248 compressive strength, and mean values were plotted. UPV is also conducted to ascertain the
249 quality and homogeneity of the mortar mixes as per ASTM C-597. Samples for XRD, TGA,
250 and FTIR tests were obtained from the crushed cement pastes of 7 days curing age and ground
251 to below 75 µm using a mortar pestle. All the samples were treated with isopropanol and diethyl
252 ether to stop the hydration process.

253 **3. Results and Discussions**

254 **3.1 Characterization of AB**

255 **3.1.1 Yield, Proximate analysis, Ultimate analysis of AB**

256 Table 3 illustrates the significant impact of pyrolysis temperature on the properties of
257 AB. As the pyrolysis temperature increases, the yield of AB decreases due to the volatilization
258 of organic constituents in biomass. This results in a yield drop from 49 to 32%, as shown in
259 Table 3. The proximate analysis reveals that increasing the pyrolysis temperature reduces the
260 MC of AB samples due to carbonization (Elnour et al., 2019). Furthermore, the AC and FC of

261 all the AB samples increase with higher temperatures (Ross et al., 2008). The ultimate analysis
262 demonstrates that carbonization increases the amount of C while decreasing the N, H, S, and
263 O (by difference). The atomic ratios (H/C) and (O/C) for the AB samples ranged from 0.024 to
264 0.014 and 0.20 to 0.14, respectively. These changes indicate greater biomass pyrolysis and less
265 hydrophilic AB surfaces at high temperatures (Elnour et al., 2019). These ratios suggest that
266 all the AB samples fall within type IV on the Van Krevelen diagram (Ollivier et al., 2022),
267 indicating significant aromaticity and high stability of AB

268 . The observed changes in ultimate analysis and atomic ratios reflect the formation of
269 aromatic and graphitic structures, which can influence the interface interactions between filler
270 and matrix (Elnour et al., 2019).

271 **3.1.2 Physicochemical analysis of AB**

272 The surface area and pore volume of biochar are critical quality indicators, significantly
273 influencing the mechanical interlocking between filler and matrix. As shown in Table 4,
274 increasing pyrolysis temperature leads to substantial improvement in both surface area and pore
275 volume. The surface area increased from 4.34 to 112.38 m²/g, while the pore volume improved
276 from 0.006 to 0.07 cm³/g. This enhancement is attributed to the elevated temperatures
277 facilitating the release of volatile organic compounds and water, resulting in a more porous
278 structure with a greater surface area (Yuan et al., 2015).

279 Fig. 7 presents the particle size distribution of all the AB samples, and the results are
280 presented in Table 4. The size of AB is determined by DLS in the following order: AB 300 >
281 AB 400 > AB 500. Compared to AB 300, the D50 and D90 of AB 400 and AB 500 are 22 –
282 45% and 38-88% finer, respectively. The size reduction is due to the increased pyrolysis
283 temperature, resulting in a material breakdown and producing finer particles (Wang et al.,
284 2013). The utilization of fine biochar enhances packing density and composite performance by
285 effectively filling microscopic pores (Kua, 2024).

286 The zeta potential measurement evaluates the surface charge of particles and is shown
287 in Table 4. The results indicate that the zeta potential of AB decreases with increasing pyrolysis
288 temperature. This negative zeta potential generates a strong repulsive force between similarly
289 charged particles, effectively mitigating particle aggregation and enhancing dispersion stability
290 (Tan et al., 2020b).

291 The pH of biochar is significantly influenced by both feedstock type and pyrolysis
292 temperature. The pH value of AB increases from 9.14 to 10.29, with increasing pyrolysis
293 temperature, indicating an alkaline nature. This pH increase is primarily due to the removal of
294 alkali salts from organic materials at higher pyrolysis temperatures (Wang et al., 2013).

295 The salinity of the biochar samples can be assessed by measuring the conductivity of
296 the solution (Gujre et al., 2022). The results presented in Table 4 indicate that the electrical
297 conductivity of AB increases with an increase in pyrolysis temperature, varying from 2.33 to
298 3.94 mS/cm. These findings agree with the previous studies (Gogoi et al., 2017), and are due
299 to the increased ash concentration resulting from the loss of volatile matter during pyrolysis
300 (Singh et al., 2017).

301 **3.1.3 Surface Analysis of AB**

302 Fig. 8 illustrates the morphology of AB particles and surface pores produced at 300°C, 400°C,
303 and 500°C. SEM analysis reveals that all AB particles exhibit ridges and oval-shaped
304 honeycomb pore structures on their surfaces (circled in yellow), reflecting the biological
305 capillary structure of the areca nut husk feedstock. The pores of varying sizes are formed due
306 to the release of volatiles and organic matter during pyrolysis (Gupta et al., 2018b). AB
307 prepared at 500°C demonstrates more closely spaced pores compared to those prepared at
308 300°C and 400°C. The ridges on the AB surface enhance surface roughness, potentially leading
309 to stronger bonding with the cementitious matrix (Gupta et al., 2018a). These pores serve a
310 dual function, absorbing water and facilitating internal curing by releasing water during the
311 hardening stage.

312

Table 3 Chemical characteristics of biochar samples

Sample	Yield (%)	Proximate Analysis (%)				Ultimate analysis (%)					Atomic ratios	
		MC	VM	AC	FC	C	H	S	N	O (By difference)	H/C	O/C
AB 300	49	4.61	16.25	4.25	74.89	78.98	1.91	2.23	0.84	16.04	0.024	0.20
AB 400	39	3.21	14.89	5.98	75.92	83.56	1.65	1.00	0.19	13.60	0.019	0.16
AB 500	32	2.46	12.66	6.52	78.36	85.12	1.20	0.85	0.14	12.69	0.014	0.14

313

Table 4 Physical characteristics of biochar samples

Sample	Particle Size (nm)			pH	Conductivity (mS/cm)	Surface area (m ² /g)	Total pore volume (cm ³ /g)	Mean pore radius (nm)	Mean zeta potential (mV)
	D10	D50	D90						
AB 300	300	1418	12601	9.14	2.33	4.34	0.006	3.01	-19.59
AB 400	185	1113	1934	9.89	2.87	44.07	0.03	1.32	-11.38
AB 500	181	783	1444	10.29	3.94	112.38	0.07	1.29	-9.64

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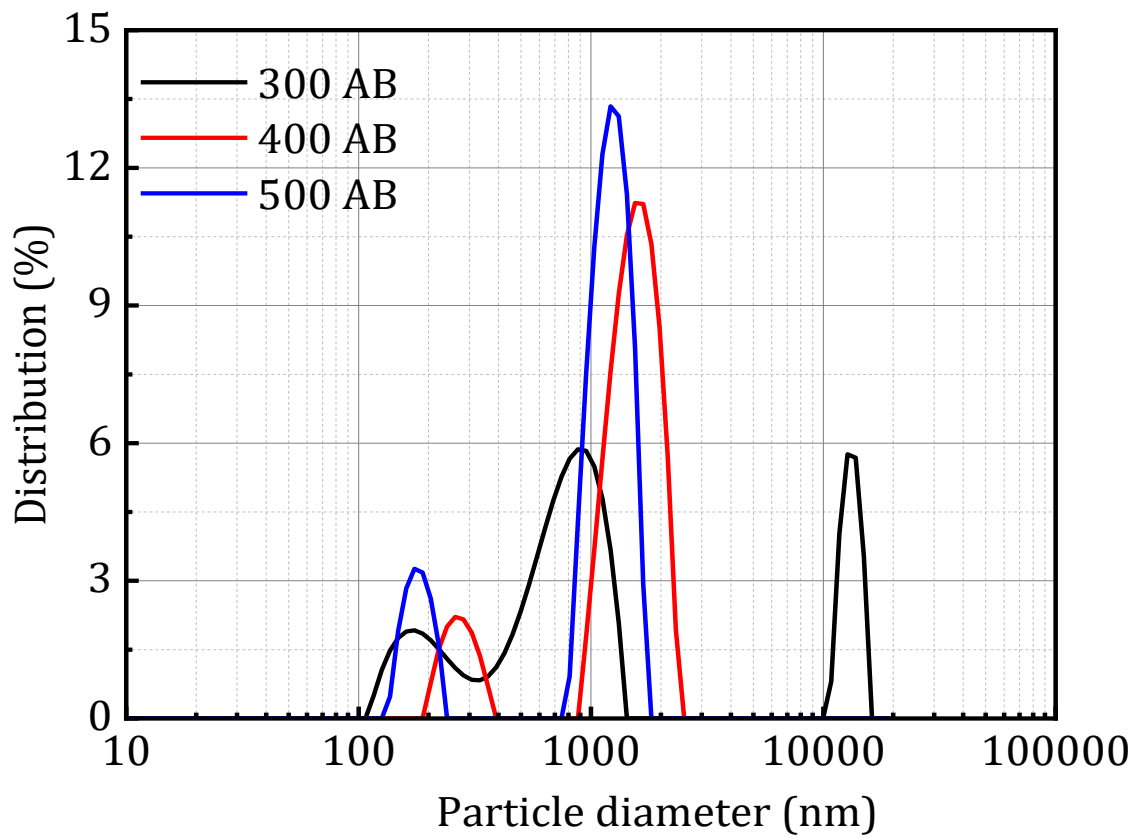
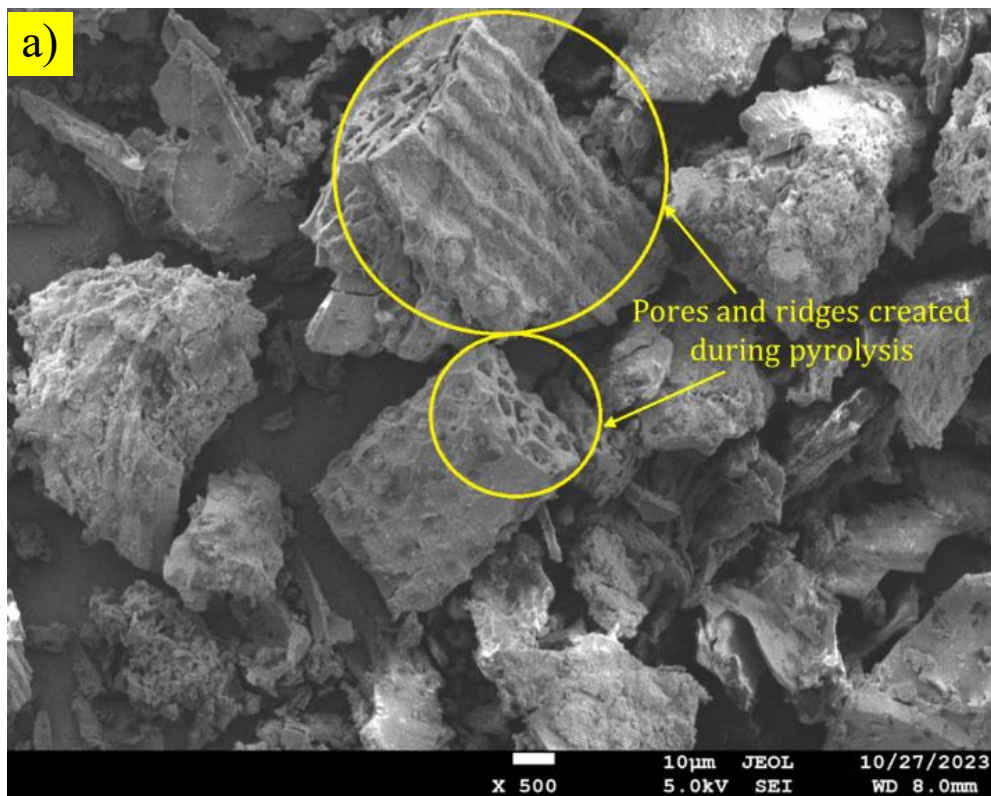
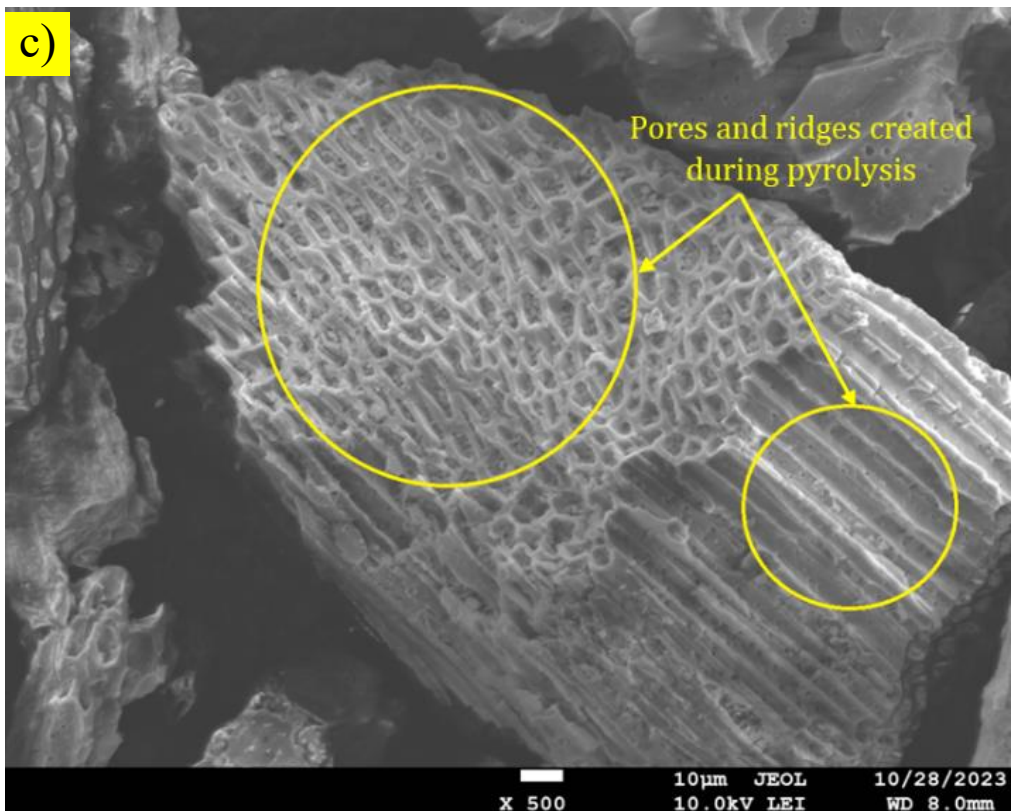
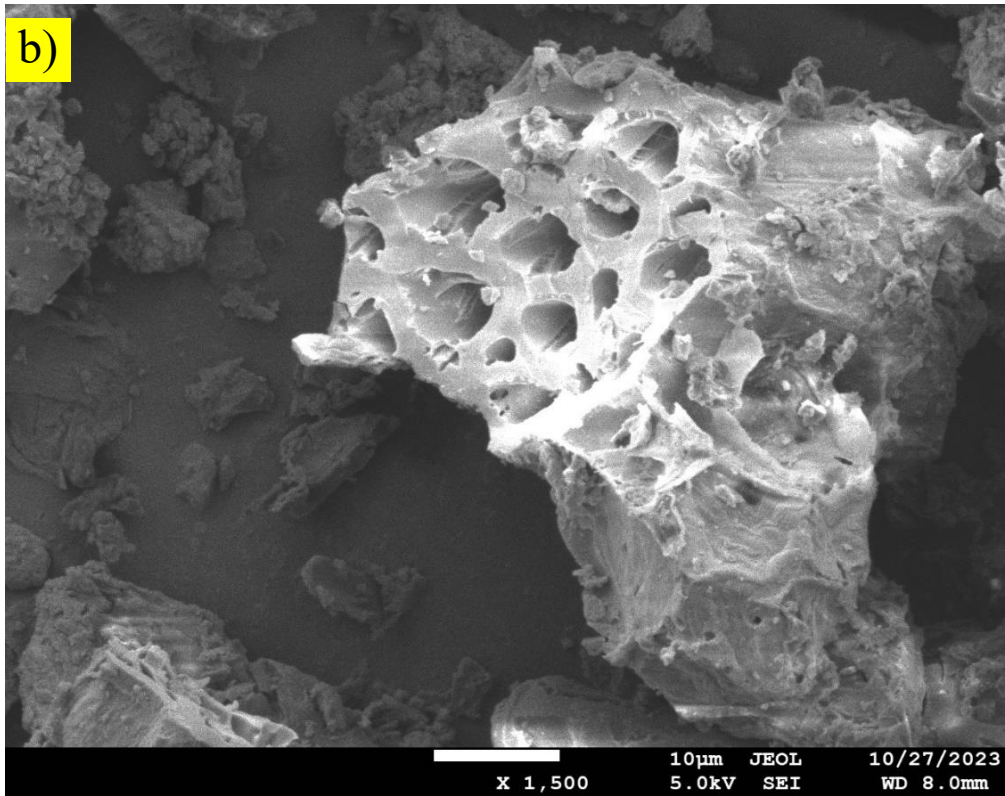
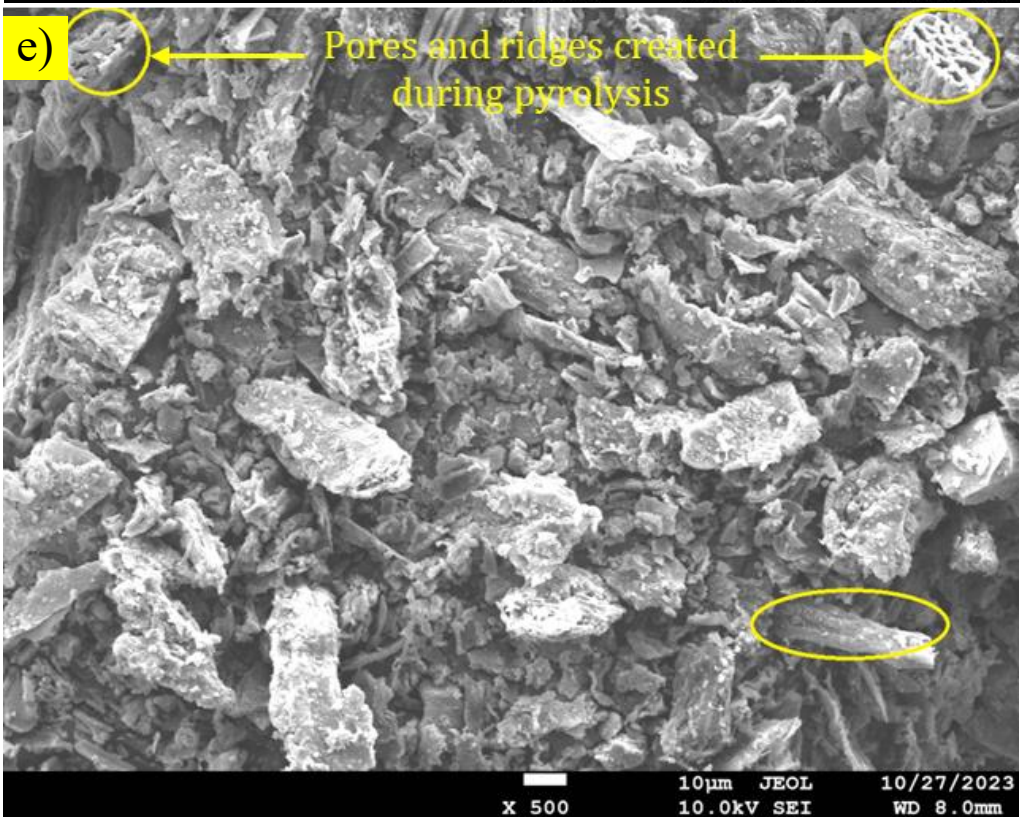
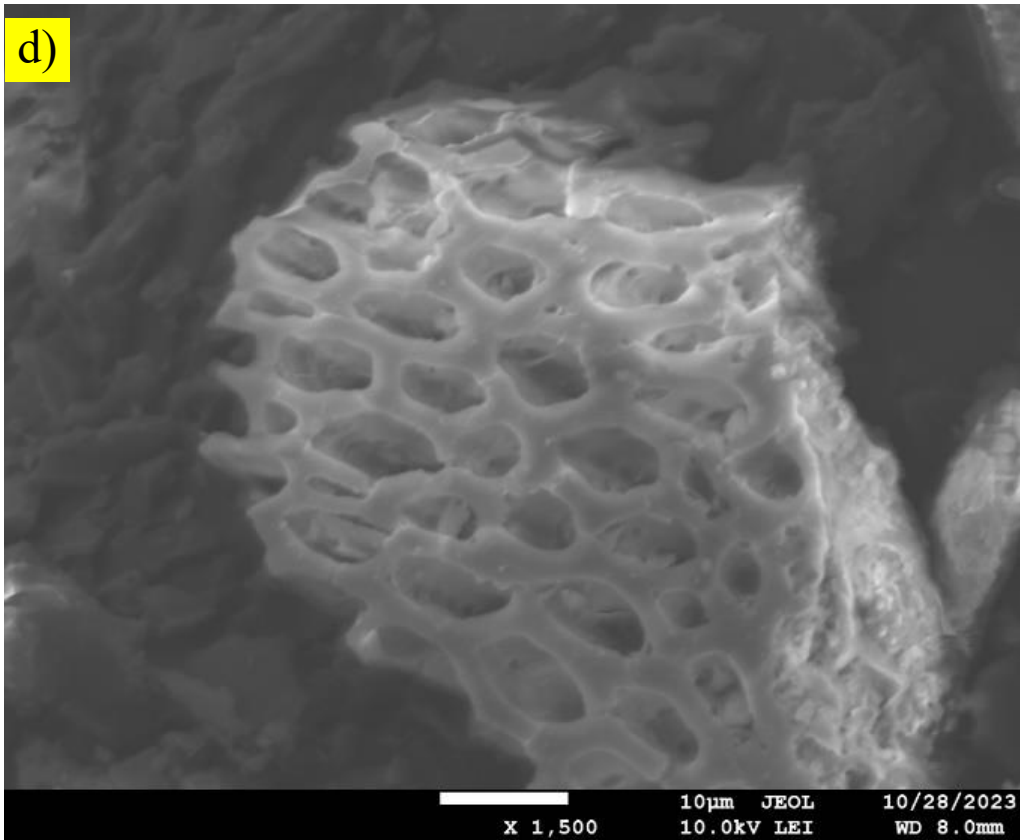


Fig. 7. Particle size distribution of AB

315







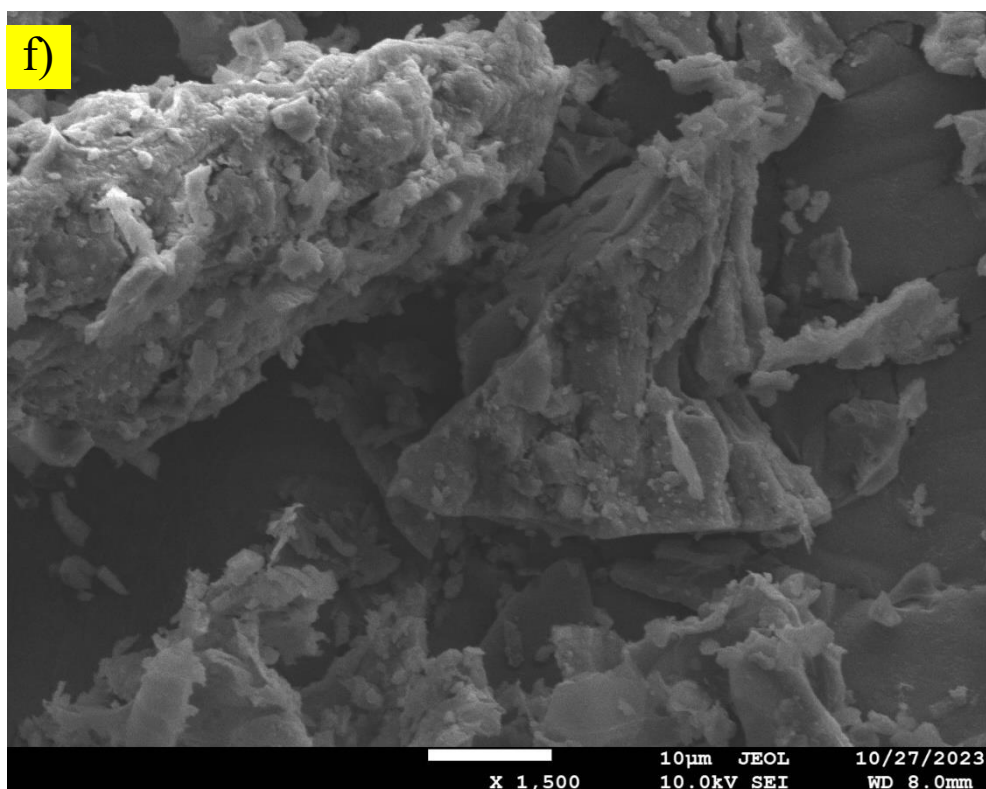


Fig. 8. SEM images of magnifications 500X and 1500X: a) and b) AB 300, c) and d) AB 400, e) and f) AB 500

316 **3.1.4 Structural Analysis of AB**

317 The thermal behaviour of AB prepared at different temperatures is studied using TGA
 318 and derivative thermogravimetry (DTG) curves. Fig. 9 shows that AB 300 experienced the
 319 highest weight loss (44%), followed by AB 400 (36%) and AB 500 (28%), with decreased
 320 weight loss at higher temperatures attributed to dehydrogenation, aromatization, and inorganic
 321 element degradation (Reza et al., 2020). The DTG curves present two prominent peaks for all
 322 AB samples. The first thermal peak phase occurs between 25 - 150°C for AB 300 and 25-200°C
 323 for AB 400 and AB 500, primarily results from moisture removal due to the hygroscopic nature
 324 of the biomass, resulting in the removal of moisture from its surface and pores (Pariyar et al.,
 325 2020). The second thermal peak phase observed between 150-650°C for AB 300 and 200-
 326 900°C for AB 400 and AB 500 corresponds to the degradation of hemicellulose, cellulose, and
 327 the gradual breakdown of lignin (Patwa et al., 2022). As pyrolysis temperature increased, the
 328 peaks for AB 400 and AB 500 shifted towards higher pyrolysis temperatures, with a significant
 329 reduction in peak height, suggesting the formation of more thermostable functional groups (Li
 330 and Chen, 2018). Similar results were observed for bamboo waste-derived biochar (Zhang et
 331 al., 2022). The thermal stability of AB prepared at higher temperatures agrees with the lower

332 atomic ratios. Table 5 presents the weight loss for various phases of the AB samples, providing
 333 an overview of the thermal degradation behaviour of AB.

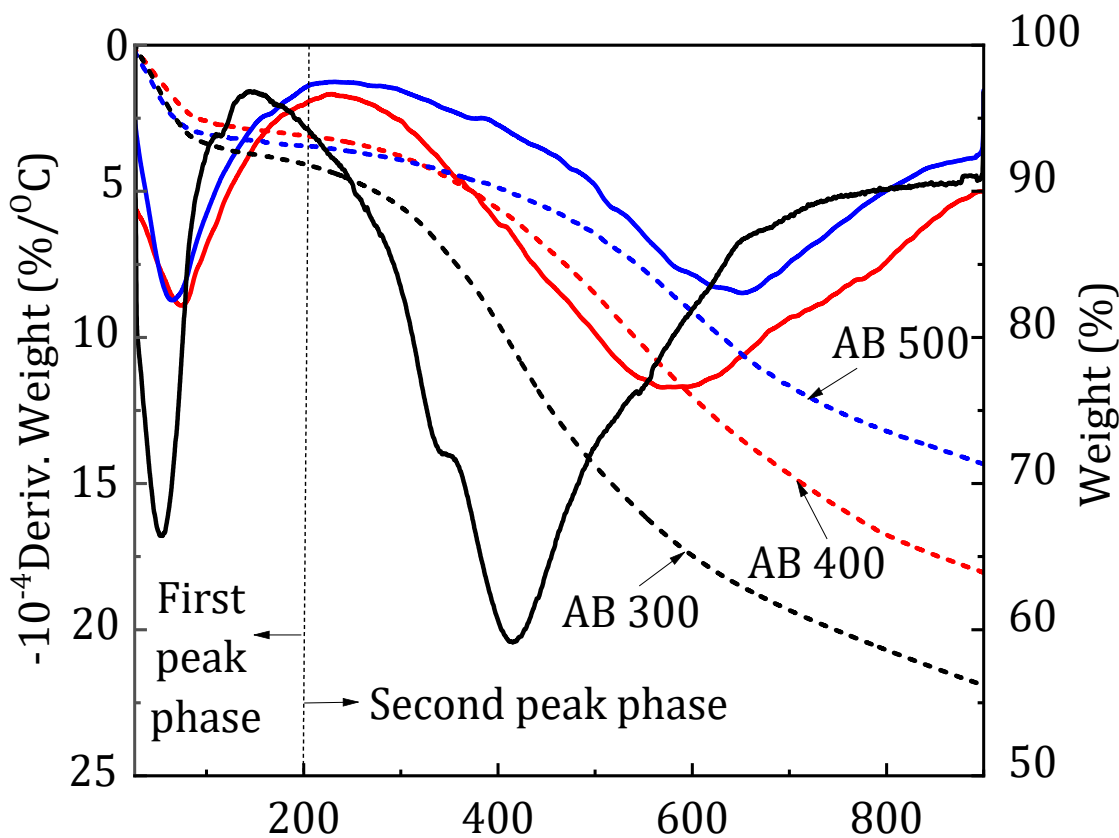


Fig. 9. TGA of AB

334 **Table 5** Various phases of the TGA and DTG for AB with weight loss

Biochar	Weight loss for stage 1 (%)	Weight loss for stage 2 (%)	Residue (%)
AB 300	7.48	39.27	53.23
AB 400	6.21	31.87	61.91
AB 500	6.89	23.39	69.71

335 Fig. 10 presents the FTIR spectra of AB produced at various temperatures. The analysis shows
 336 that the functional groups are decreased with increasing pyrolysis temperatures due to
 337 enhanced carbonization and removal of volatile matter (Rafiq et al., 2016). A peak at
 338 3388 cm^{-1} in the FTIR spectra corresponds to the O-H bond stretching of phenolic hydroxyl
 339 groups, which decreases at higher temperatures due to the loss of hydrogen and oxygen (Elnour
 340 et al., 2019). The C-H stretching vibration at 2870 cm^{-1} indicates the presence of cellulose,
 341 hemicellulose, and lignin. The peak decreases significantly as aliphatic structures transform
 342 into aromatic structures. Carbonyl bond (C=O) stretching vibration at 1710 cm^{-1} is associated
 343 with ketones and esters (Cantrell et al., 2012). An absorption band at 1594 cm^{-1} confirms

344 phenolic groups common in lignin and aromatic compounds, indicative of double bonded
 345 hydrocarbons (C=C) stretching vibrations (Sharma et al., 2004). The band at 1110 cm⁻¹
 346 corresponds to C-O-C groups in carbonyl derivatives. The absorbance at 770 cm⁻¹ indicates
 347 aromatic C-H deformation, which is more prominent at higher temperatures (Kloss et al.,
 348 2012). The formation of graphite-like polyaromatic structures at higher temperatures results in
 349 less intense peaks (Bardalai and Mahanta, 2018). Overall, AB prepared at higher temperatures
 350 exhibits lower (H/C) and (O/C) ratios, indicating fewer surface functional groups and higher
 351 carbon content.

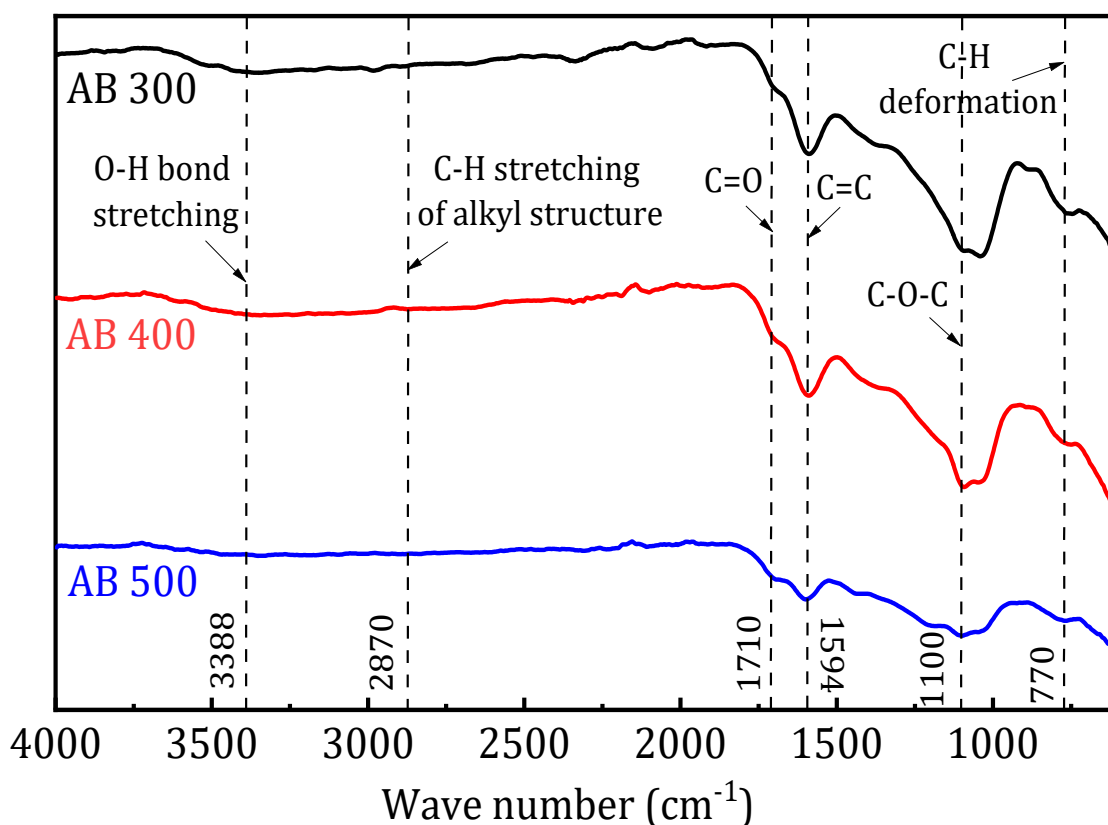


Fig. 10. FTIR Spectra of AB

352 The degree of crystallinity of AB samples was analyzed by studying the XRD, as shown
 353 in Fig. 11. The broad diffraction peak observed between 16-25°C is characteristic of the
 354 lignocellulose crystalline phase (Zhang et al., 2024), eliminated at higher pyrolysis
 355 temperatures. All AB samples exhibit a hump in the range of 16-25°, approximately 22°,
 356 indicating a carbon-rich amorphous structure (Pariyar et al., 2020) and aliphatic chains
 357 representing aromatic carbon rings. With increasing pyrolysis temperature, the peak at 26°
 358 becomes sharper and narrower, suggesting the development of highly crystalline graphitic
 359 carbon in the biochar (Kakaei et al., 2019). This observation agrees with the aromatization

360 development in the FTIR and TGA analysis. The XRD pattern of AB 500 reveals additional
361 well-defined peaks at 20°, attributed to the presence of quartz (SiO₂), which agrees with the
362 EDX of AB 500, shown in Fig. S2.

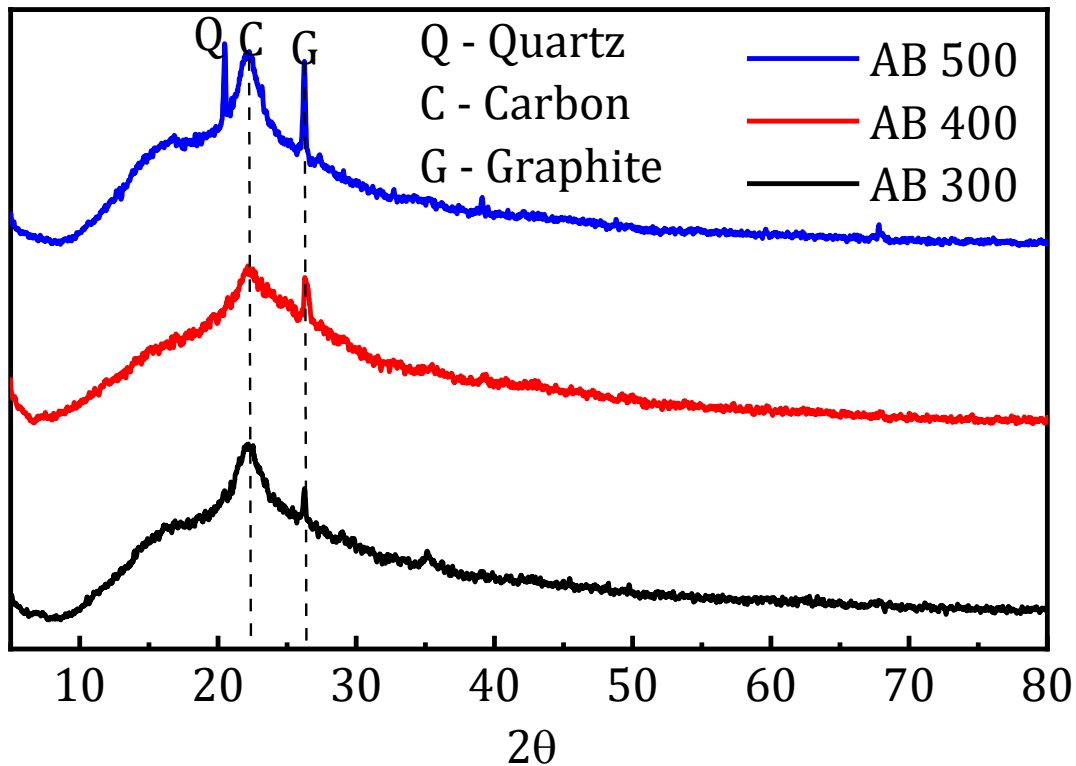


Fig. 11. XRD of AB

363 3.2 Effect of AB on Compressive Strength

364 The compressive strength of all the mortar mixes at 7 and 28 days is illustrated in Fig. 12. The
365 addition of AB significantly affects the mechanical properties of cementitious composites.
366 Regardless of its preparation temperature, incorporating AB into the mortar increases
367 compressive strength compared to the control mortar (PC). As the curing age increases, all the
368 mortar specimens consistently exhibit improved strength due to the continuous increase in the
369 hydration products. At 7 and 28 days, the compressive strength of AB 300 increased by 8% and
370 12%, AB 400 increased by 16% and 21%, and AB 500 increased by 27% and 34%, respectively,
371 with respect to the control mortar. From 7 to 28 days, the compressive strength of PC, AB 300,
372 AB 400, and AB 500 increased by 37%, 42%, 43%, and 45%, respectively. However, an
373 improvement was observed in the mortar prepared with AB 500 compared to AB 300 and AB
374 400. The reason for the improvement in strength is attributed to the micro-filler effect of AB
375 accelerating the hydration process and the fine particles of AB, resulting in a better packing
376 and denser matrix (Manjunath et al., 2024). Additionally, biochar's hydrophilic and porous
377 characteristics absorb a portion of the mixing water and subsequently release it to develop an

378 internal curing effect (Kua, 2024). This internal curing effect facilitated the production of
 379 calcium silicate hydrate (C-S-H), a critical factor contributing to strength enhancement (Javed
 380 et al., 2022). Further, the study conducted by Goldman et al. (Goldman and Bentur, 1994)
 381 reported that incorporating micro-fillers of carbon particles has a more significant
 382 strengthening effect than the pozzolanic effect.

383 The two-way ANOVA was conducted to examine the effects of pyrolysis temperature and
 384 curing age on the strength of the mortar. The results are presented in Table S3. The analysis
 385 revealed that pyrolysis temperature ($F(3,19) = 34.41, p < 0.0001$) and curing age ($F(1,19) =$
 386 $280.91, p < 0.0001$) have significant influence on the strength of biochar cementitious
 387 composites. The overall model also showed a considerable effect ($F(4, 19) = 96.03, p <$
 388 0.0001), indicating that the combination of pyrolysis temperature and curing age significantly
 389 predicts variations in the strength of the mortar.

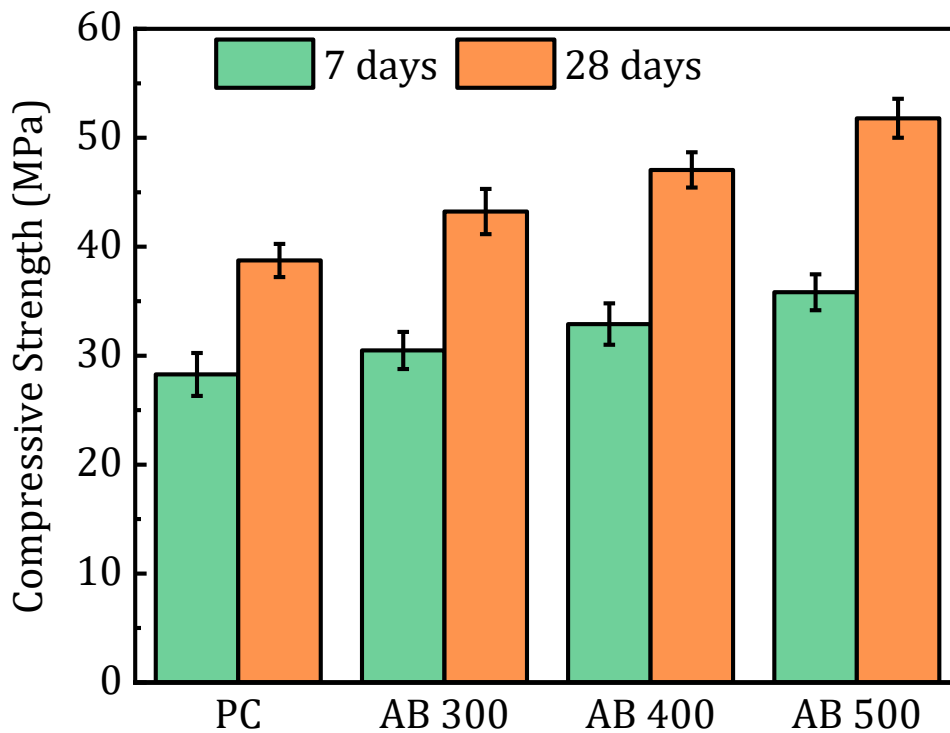


Fig. 12. Compressive strength of the AB blended mortar

390 3.3 Effect of AB on Ultrasonic Pulse Velocity (UPV)

391 The UPV is a non-destructive test used to evaluate the quality, homogeneity, and uniformity of
 392 concrete structures (Manjunath et al., 2023). It involves sending ultrasonic waves through the
 393 material and measuring the time it takes for the waves to travel from one point to another. Fig.
 394 13 presents the UPV test results for all the mortar specimens. Similar to the increase in
 395 compressive strength, the UPV of the specimens also shows an increasing trend with age. The

396 UPV of specimen PC were 4214 m/s and 4456 m/s at 7 and 28 days, respectively. The UPV of
 397 AB 300, AB 400, and AB 500 at 7 and 28 days were 4421 m/s and 4625 m/s, 4512 m/s and
 398 4708 m/s, 4630 m/s and 4810 m/s. The UPV test results for all the specimens were greater than
 399 4200 m/s, indicating good quality. However, for biochar blended samples, the UPV results were
 400 greater than 4500 m/s at 28 days, indicating its excellent quality (Research and Feldman, 1977).
 401 Biochar particles are generally finer than the average size of cement grains, and these fine
 402 particles play a crucial role in obstructing the pores within the mortar, thereby promoting the
 403 compactness of the blended mortar (Maljaee et al., 2021a). Further, the excellent water
 404 retention capacity of AB facilitates the gradual release of the retained water during later stages,
 405 effectively serving as an internal curing agent that contributes to the densification of the matrix
 406 (Gupta et al., 2018b). Consequently, this results in a shorter transmission time for ultrasonic
 407 waves to travel, leading to an increase in wave velocity.

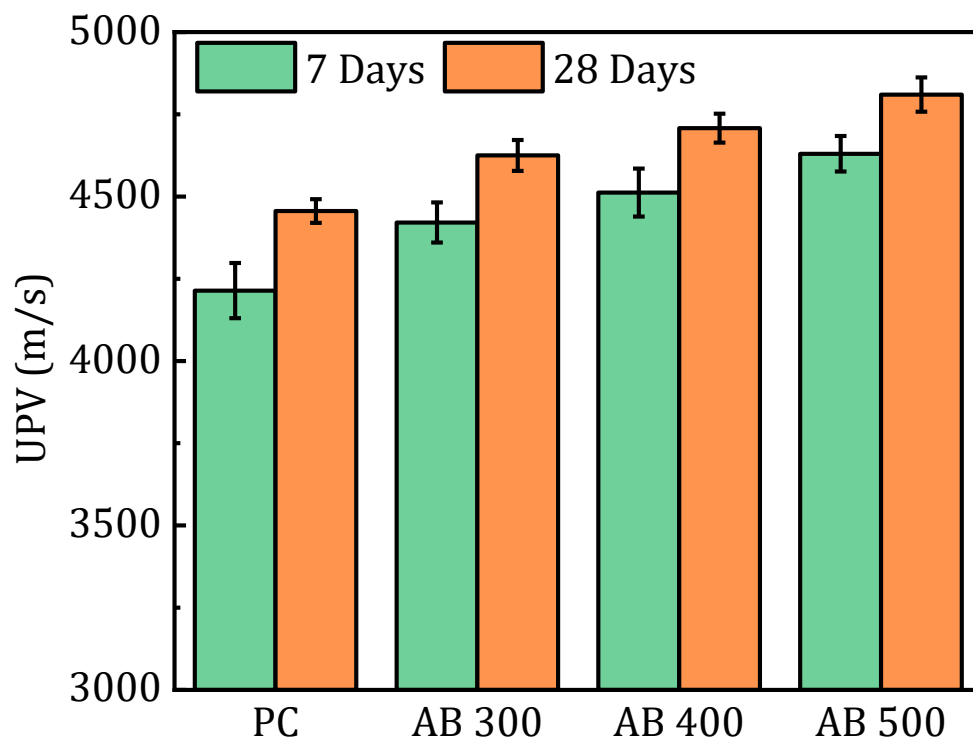


Fig. 13. UPV of the AB blended mortar

408 3.4 Effect of AB on cement hydration

409 3.4.1 Thermogravimetric analysis (TGA)

410 Fig. 14 (a) and Fig. 14 (b) show the TGA and DTG curves for the AB blended samples at 7
 411 days, and the temperature ranges from 30 to 900°C. The DTG curves are divided into three
 412 major peaks; the first peak before 100°C is due to the dehydration of C-S-H, AFt, and other

413 hydration products. This peak increases with the addition of AB, irrespective of temperature,
 414 due to the ability of the biochar to absorb free water from the surrounding environment. The
 415 second peak in the DTG curve arises from the dehydroxylation of calcium hydroxide (CH).
 416 Except for AB 300, all other samples exhibit higher peaks. This phenomenon can be attributed
 417 to the unique hydration characteristics induced by biochar. Biochar accelerates the early-stage
 418 hydration process, resulting in the formation of more CH. The third peak corresponds to the
 419 decomposition of calcite. All the AB samples showed increased peaks compared to the control
 420 sample. The accelerated hydration due to biochar leads to higher CH concentrations, leading
 421 to an increase in the calcium carbonate (CC) during carbonation.

422 TGA provides valuable insights into the distribution of hydration products. The
 423 following equation was used to determine the hydration and carbonated product formation of
 424 the AB blended cement paste based on the weight loss in the TGA.

$$425 \quad \% \text{CH}_{dx} = 4.11 \times dx_{(400-500^{\circ}\text{C})} \quad (7)$$

426 Where CH_{dx} and $dx_{(400-500^{\circ}\text{C})}$ are the calcium hydroxide content and loss of mass in the
 427 decarboxylation zone, respectively.

428 The degree of hydration (α) was determined using Bhatti's method, as outlined in the
 429 following equations.

$$430 \quad W_b = W_{dh} + W_{dx} + 0.41 W_{dc} \quad (8)$$

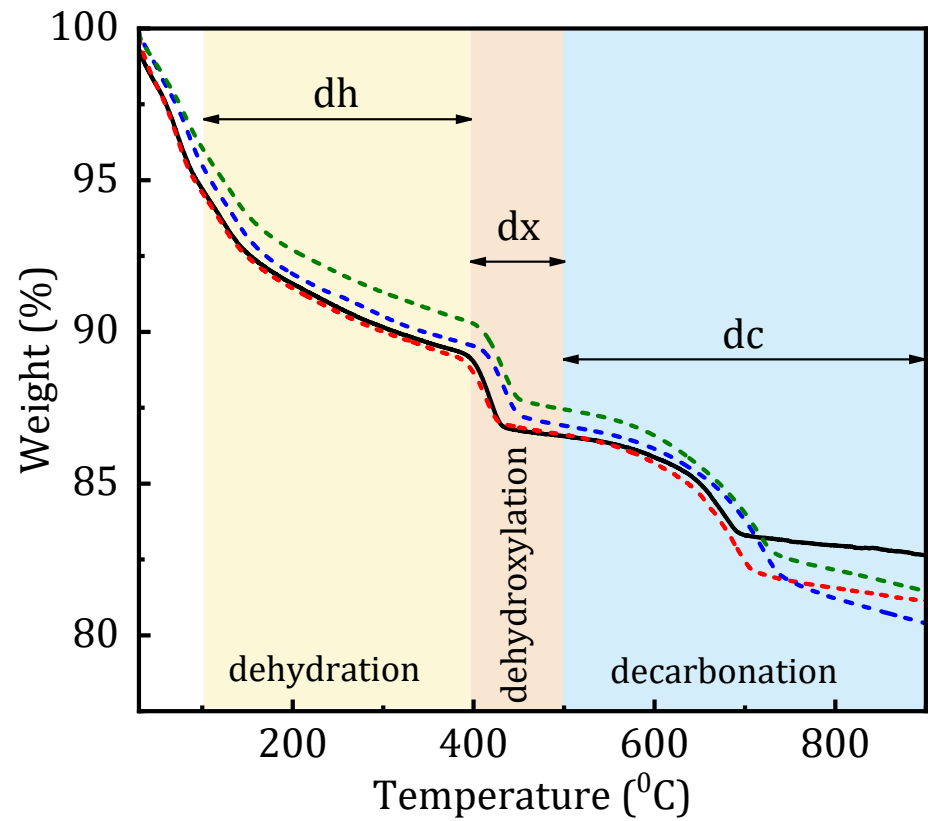
431 Where W_b = chemically bound water, W_{dh} ($105 - 400^{\circ}\text{C}$), W_{dx} ($400 - 500^{\circ}\text{C}$), and W_{dc}
 432 ($500 - 900^{\circ}\text{C}$) are the weight loss caused by dehydration, de-hydroxylation, and decarbonation.
 433 The factor 0.41 is employed to convert the mass loss resulting from decarbonation into the
 434 equivalent molecular weight of water.

$$435 \quad \alpha (\%) = W_b / 0.24 \times 100 \quad (9)$$

436 The value 0.24 corresponds to the theoretical maximum water needed for the complete
 437 hydration of cement particles.

438 The degree of hydration at 7 days was calculated based on the above equations. The AB
 439 blended samples had a comparable degree of hydration. The results from Table 6 indicated that
 440 AB blended pastes exhibited a higher degree than PC. This enhancement is attributed to the
 441 large surface area of AB particles, which facilitates additional sites for chemical reactions
 442 during hydration. Additionally, the water-retention capacity of biochar, owing to its porous
 443 structure, further contributes to increased hydration. Overall, the positive influence of AB-
 444 blended samples on cement hydration leads to improved mechanical properties. The degree of
 445 hydration is consistent with other studies (Ali et al., 2023; Dixit et al., 2019).

a)



b)

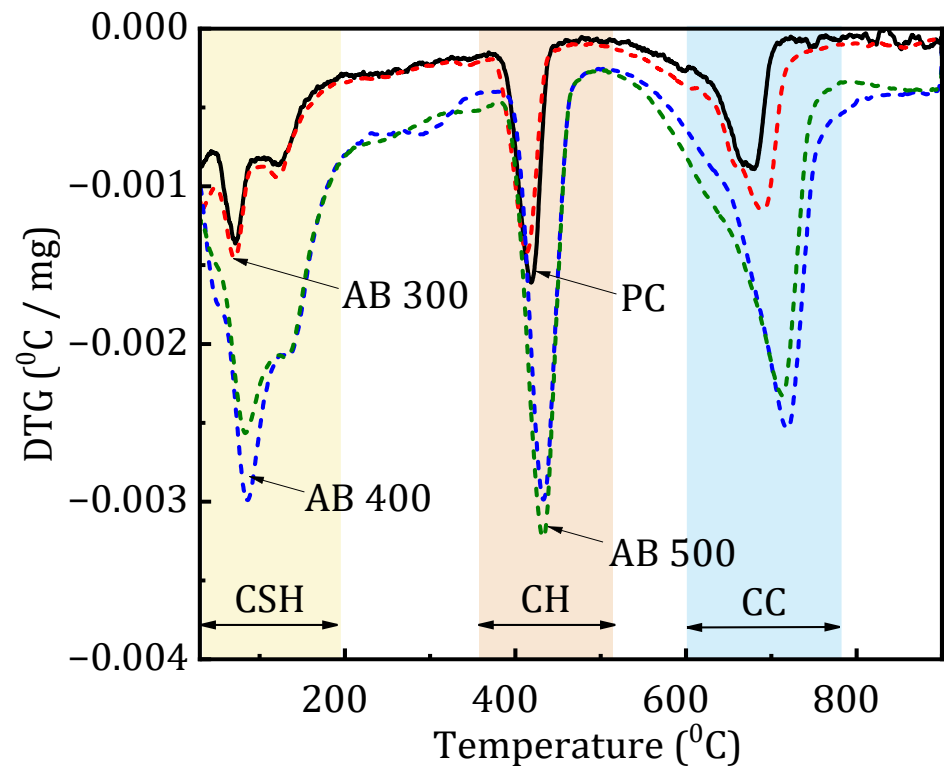


Fig. 14. TGA – DTG analysis of the blended pastes

446
447
448

Table 6 Determination of hydration products using TGA analysis.

Percentage of weight loss for hydration products at various temperature ranges							
Mixes	Biochar (%)	W_{dh}	W_{dx}	W_{dc}	W_b	CH (%)	α (%)
PC	0%	6.13	2.82	4.74	10.92	11.68	45.50
AB 300	2%	6.44	2.35	6.75	11.56	9.68	48.19
AB 400	2%	6.35	3.02	8.09	12.70	12.43	52.92
AB 500	2%	6.14	3.23	7.36	12.40	13.30	51.68

450 3.4.2 Fourier Transform Infrared Spectroscopy (FTIR)

451 The FTIR analyses illustrated in Fig. 15 were used to analyze the functional groups of control,
 452 and AB blended cement pastes at different pyrolysis temperatures at 7 days of curing. The
 453 prominent peaks are observed at 870 cm^{-1} and 952 cm^{-1} , which is attributed to Si-O-Si bonds
 454 and ν_2 of CO_3^{2-} in C-S-H, respectively (Javed et al., 2022). The intensity of the peaks suggests
 455 that AB promotes silicate phase hydration, leading to increased C-S-H formation. The ν_2 peak
 456 corresponds to the asymmetric stretching vibration of CO_3^{2-} . With the increase in pyrolysis
 457 temperature, the AB contributes to the formation of carbonates during the hydration process.
 458 The other peaks at 1110 cm^{-1} are associated with the stretching vibration of SO_4^{2-} ions in Aft
 459 (Lin et al., 2021). The peak at 3630 cm^{-1} corresponds to the stretching vibration of OH^- in CH
 460 (Ali et al., 2023). The absorption band at 1430 cm^{-1} represents the asymmetric stretching
 461 vibration of CO_3^{2-} (Hughes et al., 1995). The increased peak intensity in AB blended samples
 462 suggests accelerated hydration. This high degree of hydration contributes to the higher
 463 compressive strength. These FTIR results are in good agreement with the observed
 464 compressive strength and TGA-DTG. Incorporating AB in cement composites leads to
 465 significant hydration products, potentially enhancing the overall performance of cement
 466 composites.

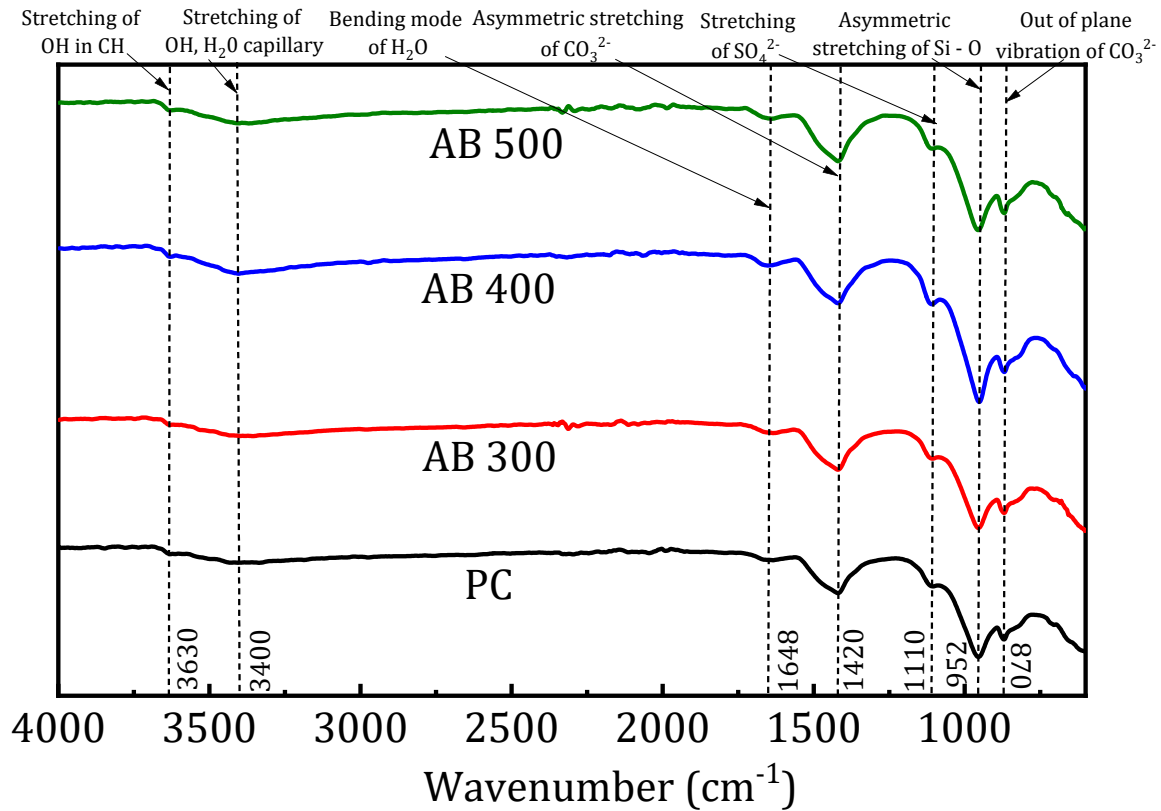


Fig. 15. FTIR analysis of the blended pastes

467 3.4.3 X-ray Diffraction

468 In the XRD results shown in Fig. 16, the cement pastes containing AB 300, AB 400, and AB
 469 500 at 7 days exhibited distinct peaks corresponding to (P – CH (calcium hydroxide), C – CC
 470 (calcium carbonate), Q – silica, T – Alite, D- Belite (Maljaee et al., 2021b). The CH peak
 471 observed at 2θ angles of 18° , 29.29° , 33.96° , 47.04° , 50.68° , and 62.38° increased in the AB
 472 blended samples, regardless of the pyrolysis temperature, than the control samples. These
 473 findings suggest that AB significantly accelerates the hydration process. Biochar particles act
 474 as a nucleation site for CH and other hydration products, resulting in higher CH concentration
 475 during the early stages of hydration (Qu et al., 2024; Zhu et al., 2023). Additionally, the CC
 476 peaks primarily detected at 2θ angles of 23° and 39.3° exhibited an increasing trend in biochar
 477 blended samples. The enhanced CH content during early hydration contributes to carbonation
 478 to form CC peaks. The results from TGA further support this observation. Unlike crystalline
 479 phases, C-S-H has a gel structure crucial for cement strength and does not produce distinct
 480 peaks in XRD. Peaks at 2θ angles 32° and 41° correspond to T and D, respectively. The peak
 481 at 26.8° in the AB 500 sample indicates the presence of Q, resulting in enhanced compressive
 482 strength. Overall, the addition of AB, regardless of the pyrolysis temperature, influences

483 hydration, promoting T and D phases and contributing to increased C-S-H content, which
484 enhances overall strength development (Javed et al., 2022).

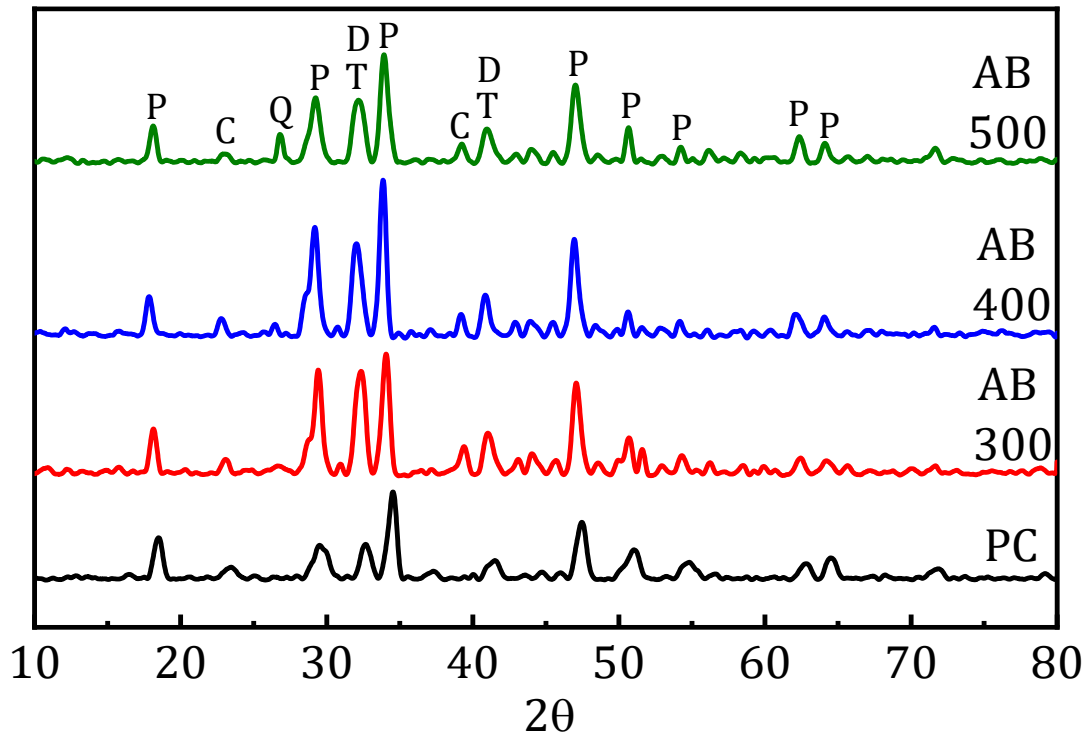


Fig. 16. XRD analysis of the blended pastes

485 Fig. 17 illustrates the hydration mechanism of cement and AB blended pastes. The
486 comprehensive analysis of the effect of AB on cement hydration through TGA, FTIR, and XRD
487 analysis findings reveals a consistent pattern of enhanced hydration reaction. The hydration
488 process results in the formation of C-S-H and CH, creating a layer of hydration products on the
489 cement grains (illustrated by blue circles), as shown in Fig. 17 (A). Fig. 17 (B) shows small-
490 sized AB particles dispersed between the larger cement grains. During hydration, AB
491 accelerates early-stage hydration and attracts positively charged hydrated cement particles
492 around them due to its surface negative charge (Gupta et al., 2021). This forms nucleation
493 clusters that promote the growth of hydration products on the surface, which is consistent with
494 the TGA and XRD results showing accelerated early-stage hydration and increased CH peaks.
495 Further, the formation of hydration products is consistent with the increased intensity observed
496 in FTIR analysis, while XRD results show higher CH and CC in AB blended pastes. This also
497 correlates with the improved mechanical properties and a higher degree of hydration observed
498 in TGA for AB-blended cementitious composites. At 7 days, the hydration process results in
499 the formation of C-S-H, CH, and CC, leading to a denser structure and enhancing the overall
500 performance of the cement composite.

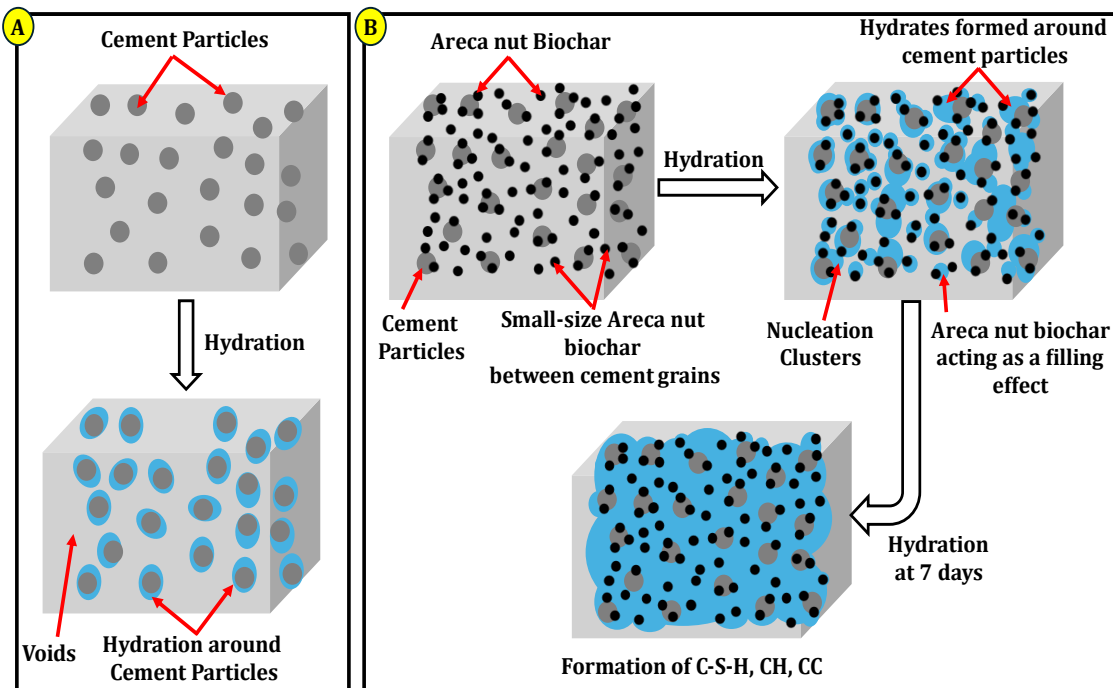
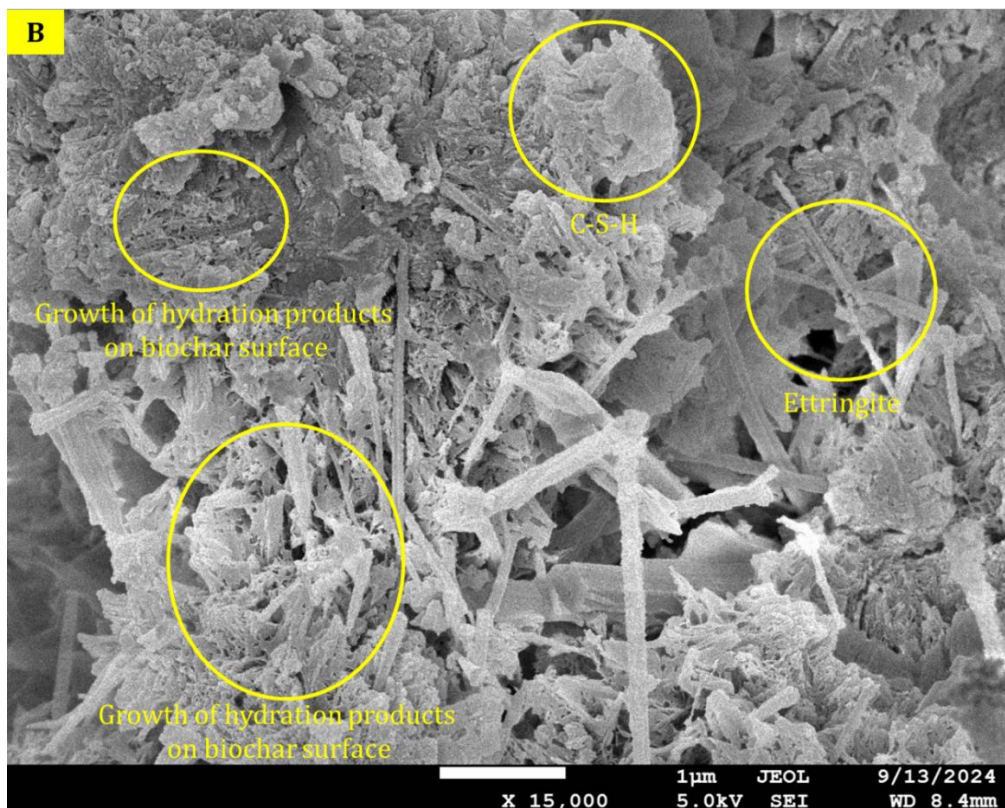
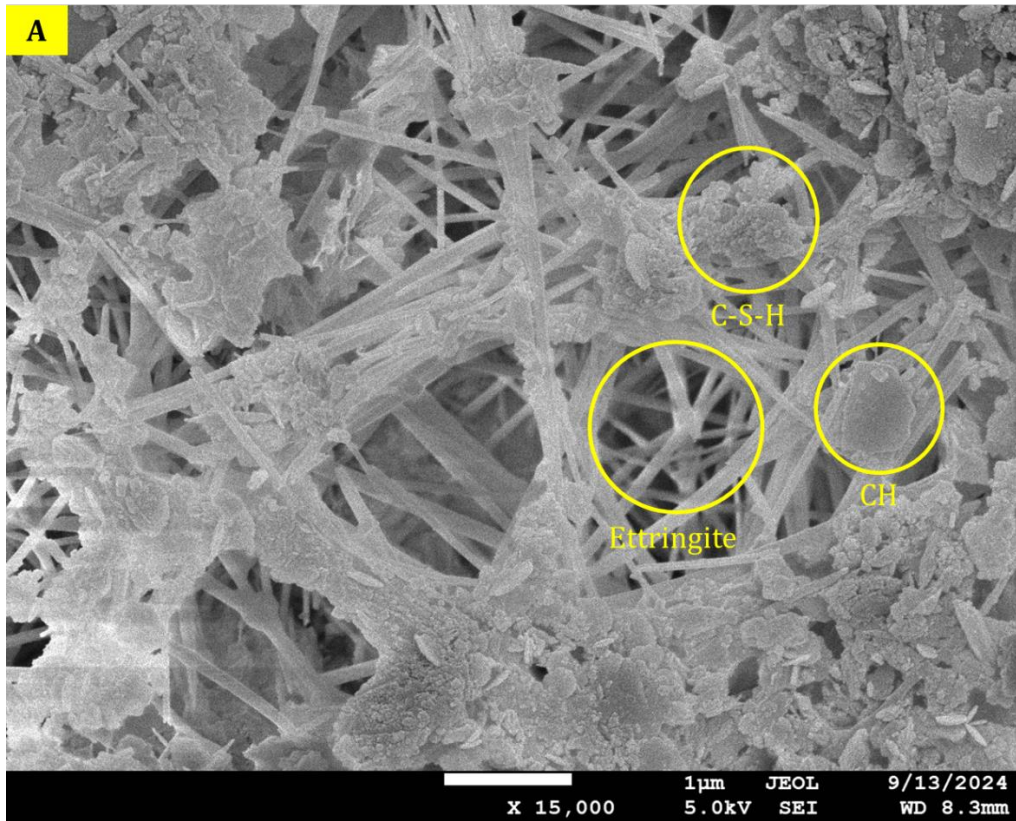
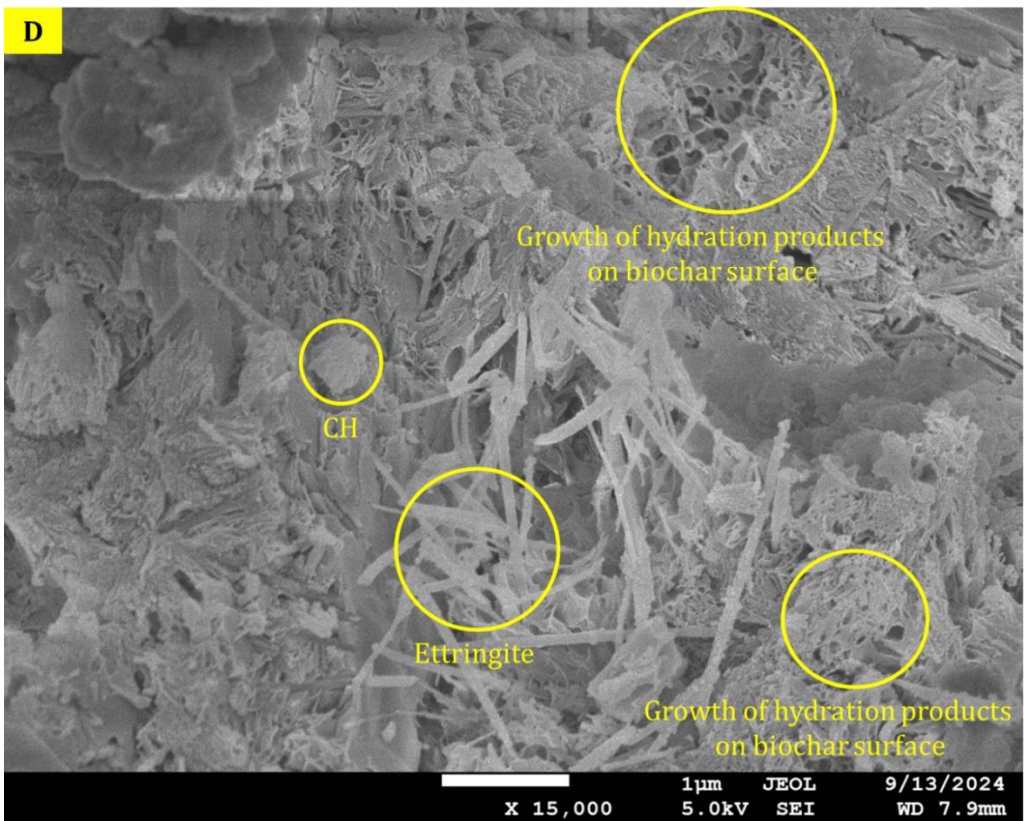
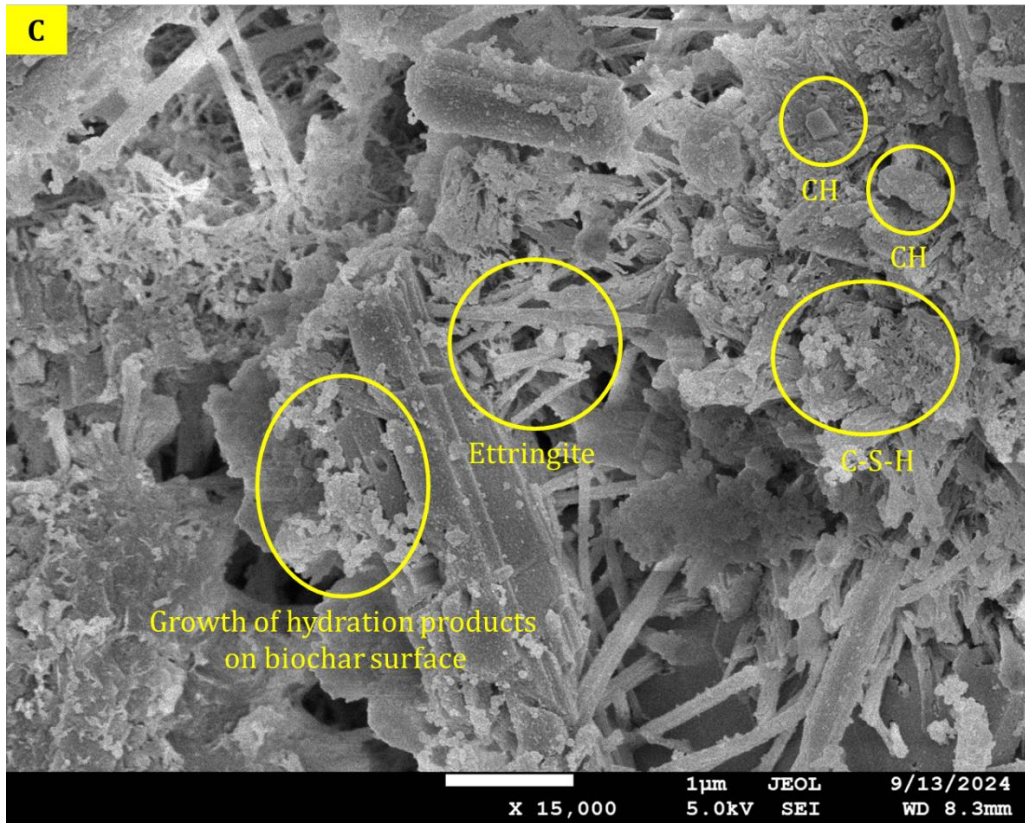


Fig. 17. Schematic diagram showing hydration product in cement and blended biochar pastes

501 Fig. 18 shows the SEM images of the control and AB blended paste (PC, AB 300, AB 400, and
 502 AB 500) after 7 days of curing. All the AB blended samples exhibited the formation of
 503 hydration products, needle-shaped ettringite, C-S-H, and crystals of CH (shown by yellow
 504 circles). Compared to PC, even a small amount of biochar provides additional hydration sites
 505 and creates a closer interconnection. Moreover, the particle size of the biochar enables them to
 506 act as micro-reinforcers in cement, resulting in denser structures. Biochar pores absorb water
 507 during mixing, facilitating the formation of hydration products within the structure, which
 508 contributes to the strength of the cement matrix.





509 **Fig. 18.** SEM image of cement and blended biochar paste A) PC B) AB 300 C) AB 400 D)

510

AB 500

511

512 **4. Environmental Benefits**

513 The environmental benefits of incorporating AB as a carbonaceous filler to cement can be
 514 measured by the net reduction in carbon equivalent emissions related to AB blended mortars
 515 with the control mortar mixes. Table 7 presents the CO₂ equivalent values associated with the
 516 materials used. The CO₂ equivalent attributed to AB can be determined by calculating the
 517 percentage of carbon sequestered within the AB while accounting for CO₂ equivalent emissions
 518 generated during the pyrolysis. CO₂ equivalent emissions from sand and water are the same
 519 throughout the mixes. Hence, it is neglected from the calculations.

520 **Table 7.** CO₂ equivalent emissions of the materials used

Material	CO ₂ equivalent (kgCO ₂ /kg)
Cement	1.002 (Grant, 2015)
AB 500	-1.771*

521 By considering AB 500, the carbon content in the produced AB at 500°C is 85.12% (or 851.2
 522 per kg). 1 kg of AB 500 would lead to CO₂ sequestration of, $(44/12) \times 851.2 \approx 3121$ g or 3.121
 523 kgCO₂/kg. The ratio of (44/12) is used because CO₂ has a molecular weight of 44, while C has
 524 an atomic weight of 12. Slow pyrolysis at 450 – 500°C emits 0.407 kg CO₂/kg of dry feedstock
 525 of switchgrass (Gupta and Kashani, 2021; Roberts et al., 2010). This value was used as an
 526 approximation, considering switchgrass and areca nut husk as agricultural residues. Yield of
 527 AB 500 $\approx 32\%$ (meaning 1 kg biochar requires 3.125 kg feedstock). The emissions calculation
 528 is $3.125 \times 0.407 \approx 1.35$ kgCO₂/kg. Net CO₂ avoidance is $(3.121 - 1.35) \approx 1.771$ kgCO₂/kg.
 529 The net CO₂ avoidance is $1.771 \times 0.32 \times 1000 \approx 566.72$ kg CO₂ equivalent of emission per ton
 530 of dry feedstock. The value accounts for both the CO₂ sequestered by the AB 500 and the
 531 emissions produced during the pyrolysis process.

532 * for every kg of AB 500 produced, there is a net reduction of 1.771 kg of CO₂ in the
 533 atmosphere.

534 Table 8 illustrates the CO₂ equivalent for the control and AB 500 blended mortar mixes for a
 535 weight of 1 kg of material.

536 **Table 8.** CO₂ equivalent for the AB blended mixes

Mixes	Cement (kg)	AB (kg)	CO ₂ equivalent (kgCO ₂ /kg)
PC	1.00	0	1.002
AB 500	0.92	0.08	0.780

537 It can be observed from Table 7 that the addition of AB 500 at 2% reduces the CO₂ equivalent
538 by 22%. However, it is observed that the CO₂ equivalent of biochar varies depending on the
539 feedstock, yield, carbon content, and dosage. To maximize the potential of carbon sequestration
540 and sustainability in the construction sector, prioritizing the utilization of locally available
541 wastes for biochar production offers environmental and economic benefits. This approach
542 significantly reduces transportation-related emissions, minimizes waste sent to landfills, and
543 decreases energy consumption associated with material processing. By promoting a circular
544 economy, it transforms local waste into valuable resources, stimulating local economies and
545 reducing dependence on imported materials. This localized strategy minimizes the carbon
546 footprint of both biochar production and construction processes and enhances the net positive
547 impact on carbon sequestration and overall sustainability.

548 **5. Conclusions**

549 This study presents an innovative approach for converting waste areca nut husk into biochar as
550 a novel carbonaceous filler for cement. The conclusions of the findings are outlined below.

- 551 • The pyrolysis temperature has a significant influence on the characteristics of AB.
552 Higher temperatures accelerate carbonization and promote the formation of aromatic
553 carbon structure.
- 554 • The AB blended cement mortars showed higher compressive strength than the control
555 mortar at 7 and 28 days. The micro-filler effect and porous nature of AB accelerate the
556 hydration process by absorbing a portion of the mixing water and subsequently
557 releasing it, thereby enhancing the compressive strength.
- 558 • TGA, FTIR, XRD, and SEM analysis confirmed increased hydration product formation
559 in AB cement pastes. Furthermore, the degree of hydration from Bhatti's method
560 increased with the addition of AB.
- 561 • The partial replacement of cement with AB effectively reduces CO₂ equivalent
562 emissions in cement mortar production through decreased cement consumption and the
563 negative carbon footprint of AB.

564 AB offers a sustainable solution as a novel carbonaceous filler, enhancing strength and
565 hydration in cement composites. The findings from this study will provide a foundation for
566 further research to optimize the performance of AB and its benefits in engineering applications.
567 Finally, locally available agrowaste for biochar production presents an attractive option for the
568 future of the construction industry.

569 **Credit authorship contribution statement**

570 **Balasubramanya Manjunath:** Conceptualization, Investigation, Data curation, Validation,
571 Formal analysis, Methodology, Visualization, Writing - original draft, Writing - review &
572 editing, Funding acquisition. **Claudiane M. Ouellet-Plamondon:** Conceptualization, Data
573 curation, Validation, Writing - review & editing. **B.B. Das:** Data curation, Validation, Writing
574 - review & editing. **Subba Rao:** Data curation, Validation, Writing - review & editing.
575 **Chandrasekhar Bhojaraju:** Conceptualization, Investigation, Data curation, Validation,
576 Formal analysis, Methodology, Visualization, Supervision, Writing – review & editing. **Manu**
577 **Rao:** Conceptualization, Investigation, Data curation, Validation, Formal analysis,
578 Methodology, Visualization, Supervision, Writing – review & editing.

579 **Declaration of competing interest**

580 The authors declare that they have no known competing financial interests or personal
581 relationships that could have appeared to influence the work reported in this paper.

582 **Data availability**

583 No data was used for the research described in the article.

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